

Developing Field Emission Models Employing Nanoscale Surface Characterization

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Vacuum Arc Initiation

- We are interested in modeling a variety of discharge situations: from streamers at atmospheric pressure to vacuum arcs
- Vacuum discharge is critical to many modern devices.
 - Critical failure mechanism \rightarrow Want to avoid
 - Mode of operation \rightarrow Want to have predictable behavior
- We want to understand vacuum field emission from well-characterized surfaces to create physicsbased models for use in large-scale PIC-DSMC breakdown simulations
 - Field emission is necessary precursor to a breakdown event. No field emission \rightarrow no breakdown.
 - Employ Scanning Tunneling Microscopy (STM) and Photoemission Electron Microscopy (PEEM) to characterize surface very locally, and then apply high fields to initiate breakdown. Very locally ~0.1-10 nm.
 - Address the problem of not knowing the state prior to discharge at the location of discharge by characterizing and then discharging.
 - Apply known layers of dielectric (e.g., TiO₂, MgO) to challenge models and begin investigation of role of surface contaminants and oxide layers.
 - Utilize a "meso-scale" (0.1-1.0 μm) model of the surface for PIC-DSMC simulation of breakdown.

RF vs. DC breakdown "breakdown rate vs. conditioning" vs. "single shot probability vs. surface state". Then pulsed DC.

Vacuum Field Emission Measurements via STM

A pin-to-plane configuration in a scanning-tunneling microscope (STM) is employed. We also incorporated atomic layer deposited (ALD) surface films.





Why Local Characterization?

- Fowler-Nordheim field emission:
 - Typical used in macroscale models to curve-fit measured j(E) from the as-built electrode
 - Can result in β ~10-1000 !!!
- We want to locally characterize the surface to eliminate β as a fit parameter:
 - Use scanning tunneling microscope (STM) and/or atomic force microscopy (AFM) to measure topology at <= 10 nm resolution.
 - By meshing the microscopy surface and solving E-fields local to that surface, β comes out naturally – no need to "fudge" it.



Pixel-resolved ϕ using spectroscopic PEEM <u>PEY(*hv* sweep)</u>

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PEEM Measurement of Work Function Variation

- Measured spatial variation of local work function using photoemission electron microscopy (PEEM)
 - Variation across given Pt surface relatively small only a few percent
 - However, ϕ is in the exponential and the tail of the distribution can initiate field emission and eventually breakdown
- Significant (~10%) decrease in the work function due to surface contaminants picked up via exposure to air
- Use the ~10 nm-scale PDF's in mesoscale model to set element work functions in PIC-DSMC simulations



Model Development

- Create Pt electrode via sputter deposition
- Controllably contaminate Pt via Atomic Layer Deposition
- Measure work function, local topology, and electron emission for sample
- Generate probability density functions (PDF) for local work functions and effective topological field enhancement
- Incorporate measured *atomic-scale* distributions into discharge simulations by populating time-varying *meso-scale* element-based data from the PDFs
- Compare family of plasma discharge simulations to measured breakdown behavior

These curves PDF depend on the surface material, conditioning, etc. φ, β Large $j_{\rm e}({\rm E(t)}, \phi, \beta)$

surface mesh in the plasma code

AFM Surface Characterization

- Took the AFM (x, y, z) spatial points (here ~20 nm resolution) and map into Cubit meshing software
- Actual surface has virtually no significant topology we will see later that $\beta \sim 1$ everywhere
- To demonstrate significant spatial variation of field emission across the surface we also compute results with the surface relief multiplied by 10×

As-measured surface relief



Surface relief increased by 10×



AFM Topology \rightarrow Topological Atomic-Scale β

- Planar anode ~10 um above surface (ignore tip in pic!)
- Compute E_{norm} and A_{proj} for every element face in the resolved STM mesh
 - < 10 nm elements; ~600K surface faces</p>
- Get projection factor, $f_{proj} = \frac{\sum_{faces} A_{face}}{\sum_{faces} A_{proj,face}}$ • For present data $f_{proj} \sim 1.15$
- Create ~20 nm scale PDF of $\beta = \frac{E_{norm}}{E_{applied}}$
- Some elements will have $\beta < 1$
 - Globally the surface could be tilted
 - Sides of "sharp" atomic features



- We have measured atomic-scale (1-10 nm) PDFs of the work function and topological field enhancement factor
- Must convert these to the mesoscale (0.1-10 μm). Some options:
 - 1. Just pick the meso-scale β and ϕ from the atomic-scale PDFs
 - 2. Make an effective β and ϕ to use at the meso-scale
 - 3. "Brute force" for each meso-scale element face, pick N local emitters (unique β 's and ϕ 's)
- Option #1 obviously has artificially large variation for different surface realizations in simulations. We will not consider it further.
 - Sometimes get an extreme tail value and then field emit based on the mesoscale element's area
 - Other times there will be no tail values picked and no field emission until much higher fields
- We will proceed to do #2 and #3 and compare to resolved $\beta = 1$ everywhere.

- Can we make an effective β (and ϕ) from the data and/or atomic-scale β PDFs?
- Measure/compute the total field emission current versus E_{applied}
- Non-linear solve for β_{eff} :



- This makes sense: small β regions "turn on" at higher fields and pulls the effective β lower
- The precise functional form depends on the
 - Feng and Verboncoeur, PoP 13, 073105 (2006) Jinpu Lin et al., J. Appl. Phys. **121**, 244301 (2017)

• We are left with "brute force" -- for each mesoscale element face, pick N local emitters (randomly pick unique β 's and ϕ 's) from the atomic-scale measured distributions:

$$N = \frac{A_{element}}{A_{resolved}} f_{pro}$$

Must scale the number of local emitters to draw:



- However, we don't have to store all N local emitters for each surface element face
 - Field emission is highly nonlinear, and most emitters (β and ϕ) can be neglected
- Store every atomic-scale emitter (β and ϕ) that appreciably contributes to the current
 - A threshold current contribution of 0.1% results in storing $\sim 0.01\%$ of the atomic-scale emitters
 - 1 μ m² element has 10⁴-10⁶ atomic-scale emitters \rightarrow store < 1000 emitters.
- PIC field emission algorithm each Δt :
 - Compute E_{norm} on each surface element face
 - Loop over all ~100 atomic-scale emitters:

$$I_{face} = \sum_{emitters} A_e A_{FN} \frac{(\beta_e E_{norm})^2}{\phi_e t^2(y)} exp \left[-\frac{B_{FN} v(y) \phi_e^{1.5}}{\beta_e E_{norm}} \right]$$

Simulation of Emission from AFM Surface

- With the resolved ($\Delta x < 10$ nm) mesh, simulate the emission from the AFM surface
 - Show contours of e⁻ density just above the cathode surface
 - Some clipping of the topology is seen for the largest feature
- See several large-scale features that emit, otherwise very little emission



Simulate emission in PIC-code



Mesoscale Field Emission Simulations

- Meso-scale model does show stochastic variation in the e- density just above the surface based on the random seed
- Goal is to be able to sample many possible surfaces (e.g., different β 's and ϕ 's) and compute breakdown probabilities for as-built surfaces
- Contours of electron density just above the cathode show very different spatial variation between the meshed STM surface and the flat, meso-scale surfaces
 - The STM surface was sputtered deposited $Pt \rightarrow large$, $\sim micron-scale$ features are apparent
 - The current model picks atomic-scale emitter properties (β 's and φ 's) independently for every "meso-scale" surface elements. Clearly not independent for sputtered deposited Pt.

STM surface topology



Mesoscale ($\Delta x = 100$ nm) surface



STM ($\Delta x < 10$ nm) surface



.0e+2

1e+20

le+19

le+18

- 1e+17

0e+1c

Mesoscale Field Emission Simulations

- Compare computed global current versus applied Current (mAmps) field for the resolved STM surface and meso-scale model surface
 - Stochastic variation in the mesoscale currents small

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- The mesoscale model currents have the same trend as the STM surface, but $\sim 12 \times i_{STM}$
 - Difference partially (mostly?) from variation in fields due to changes in gap distance for the STM surface
 - Flat anode placed 10.4µm from the mean STM cathode height



Applied Field (GV/m)

Initial Local STM Breakdown Results

- Took local field emission i-V curves with tip radius < 100 nm at a distance of ~200 nm
- Relatively feature-less surface with small-β within the region of the tip field footprint
- Breakdown at ~4 GV/m!

- This seems to be evidence that, at least for relatively smooth sputter deposited Pt, we do not have small-β atomic-scale features that grow into large-β features which then allow breakdown to occur at ~10 MV/m.
- Perhaps there is a special feature somewhere on a $\sim 1 \text{ cm}^2$ electrode that results in (or can grow to) a large enough β to get breakdown at $\sim 10 \text{ MV/m}$ that was not present on our $\sim 10^{-6} \text{ cm}^2$ sampled area.

Conclusions

- Investigating surfaces at the atomic scale to characterize features relevant to vacuum field emission.
 - Surfaces that we characterized are extremely flat: $\beta \sim 1$ over 100's of μm^2
 - Want to clarify β -based field emission so β really is only geometry induced field enhancement.
- By examining field emission at the nanoscale, we have attempted to create a mesoscale physics-based model suitable for predictive (and stochastic) PIC simulation of emission
 - Still have a long way to go working on how to handle the correlation between beta and work function.
- Characterized region, then performed local discharge in STM (spatially constrained surface participation) \rightarrow Breakdown occurred at ~4 GV/m!
 - Region was flat and uninteresting the breakdown field is consistent with breakdown from region with a small β

We have a Low Temperature Plasma Research Facility to collaborate with external partners. Please see www.sandia.gov/prf.

Center for Integrated Nanotechnologies

