

# Using ALD Coatings to Cure Breakdown

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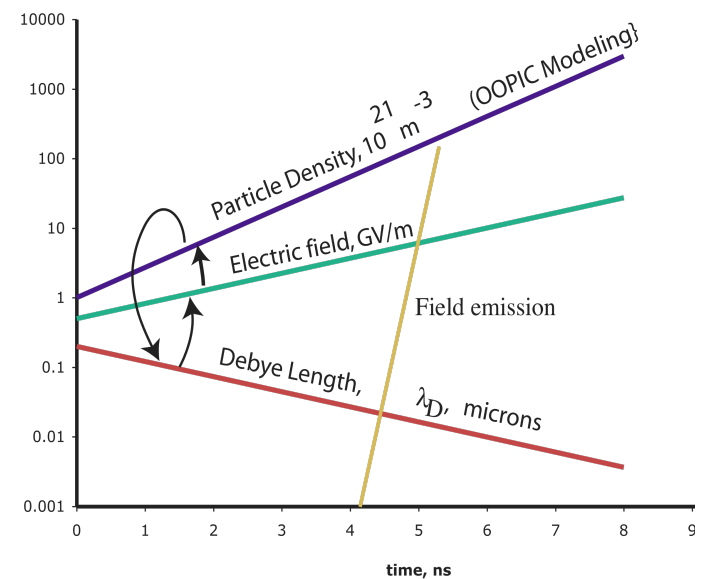
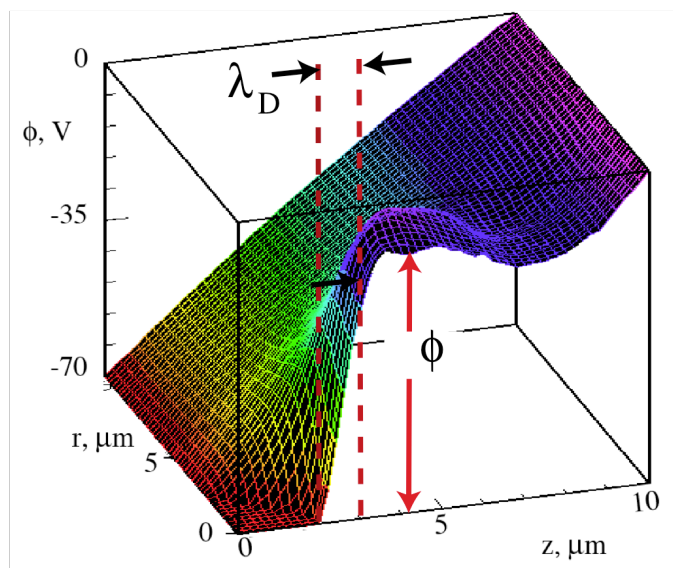


## Arc Modeling at Argonne.

- Starting in 2002, with support from ANL LDRD funds, we have been modeling arcs.
- The original aim was twofold:
  - 1) Learn how arcs worked so we could cure them
  - 2) Bring additional funding (perhaps from the ILC) into the muon effort
- The effort was not entirely successful:
  - 1) We now have a solid arc model (the first?), and a potential cure for arcing.
  - 2) We never got an additional penny from DOE for this work.
- We are refining and extending the model, which seems to apply to all accelerator, and some plasma physics problems.
- We find that a lot of other work seems to be based on misleading assumptions.  
(We've got a little list.)

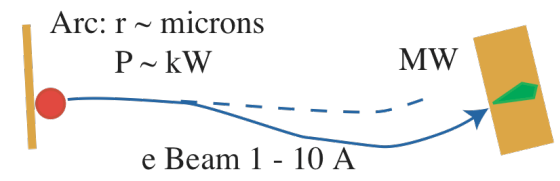
## Our breakdown model is basically simple.

- Coulomb explosions trigger breakdown - fatigue (creep) and Ohmic heating help.
- Breakdown arcs are initiated by FE ionization of fracture fragments.
- The arcs are very small, dense, cold, and charged  $+(40-100)$  V, (OOPIC and  $V_{\text{burn}}$ ).
- Small Debye lengths,  $\lambda_D = \sqrt{\frac{\epsilon_0 KT}{n_e q_e^2}} = \sim \text{nm}$ , produce fields,  $E = \phi/\lambda_D \sim \text{GV/m}$ .
- High electric fields produce micron-sized unipolar arcs.
- Unipolar arc energy produces craters and cracks with high field enhancements.



# The model describes many aspects of arcs.

## Arc Mechanisms



Fracture  
 Ohmic heating  
 Polarity dependence  
 Creep / Fatigue  
 Material dependence  
 Surface modification  
 Adsorbed gas  
 Oxides  
 Mechanical stress/strain  
 DC/rf comparisons  
 BD rate(E)  
 FE, RD emission  $I(E, \phi, T)$   
 Space charge limit  
 Thermal dependence  
 Weighted aver. of  $E_{\text{surf}}$

Microgeometry depend.  
 Coulomb Explos.  
 Evolution of ionization  
 Mass Thresholds  
 Space pot. evolution  
 Neutral gas density  
 Trapped electrons

Laser Ablation  
 e beam welding  
 Micrometeorites  
 Tokamak edges

Gas  
 Polarity

Remove E

Surface  
 Failure  
 [ $E_{\text{local}} \sim 8 \text{ GV/m}$ ]

Plasma  
 Initiation  
 [ $N_{\text{atoms}} \sim 0.5 \text{ monolayer}$ ]

"Exponential"  
 plasma growth  
 [Available energy]

Stored energy  
 Frequency dependence  
 Fueling  
 self sputtering etc  
 temp dependence  
 ion wall heating  
 line radiation heating  
 ohmic heating.  
 Magnetic fields  
 Ion etching  
 Explos. Elect. Emis,  
 Plasma growth times  
 Cavity discharge time  
 Cavity discharge current  
 how it is absorbed  
 interactions with B  
 Space charge limits  
 Liquid surface stability  
 Particulate generation  
 Unipolar arcs  
 Arc electrons to wall

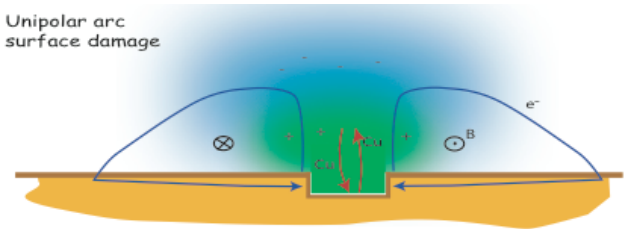
Gradient  
 Limits

Surface  
 Damage  
 [ $s(\beta) \sim \exp(-0.03 \beta)$ ]

Change B, p, t

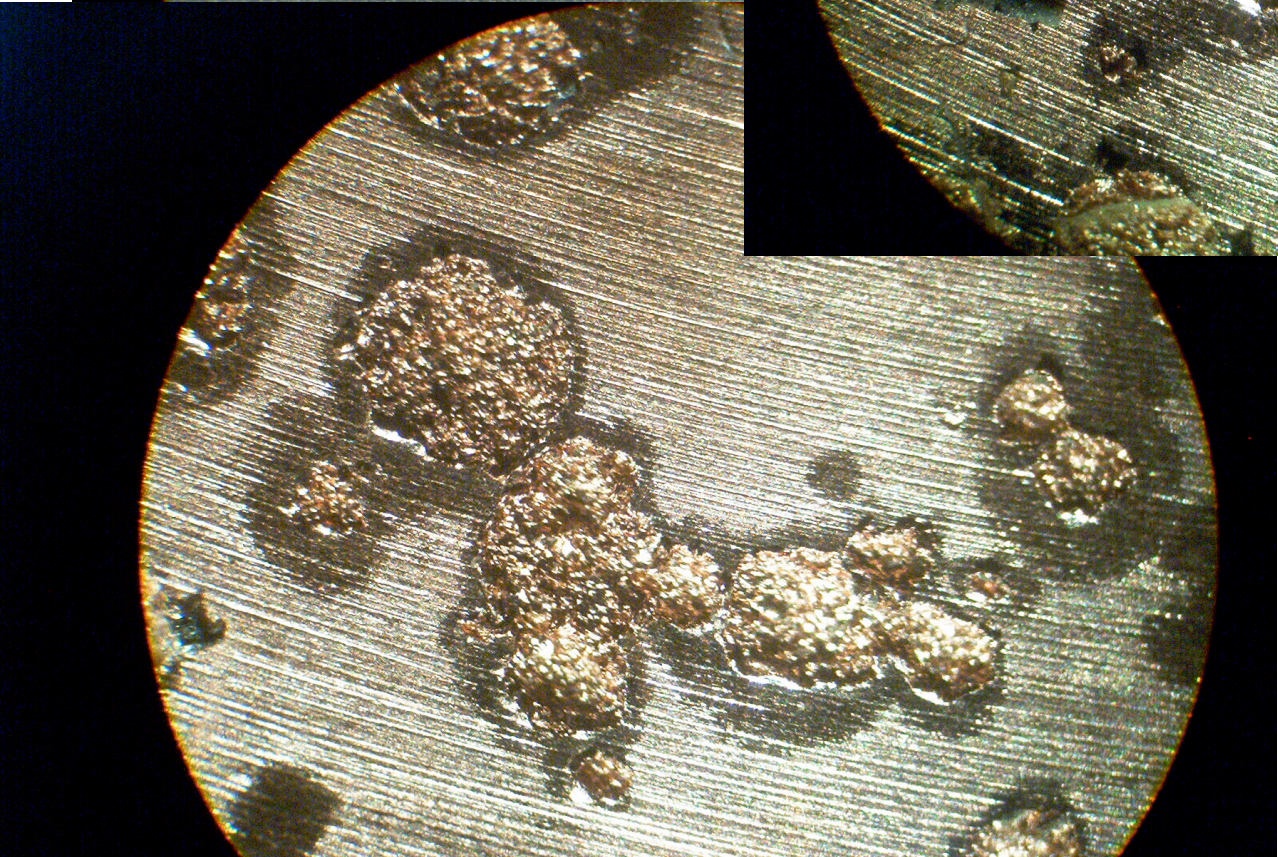
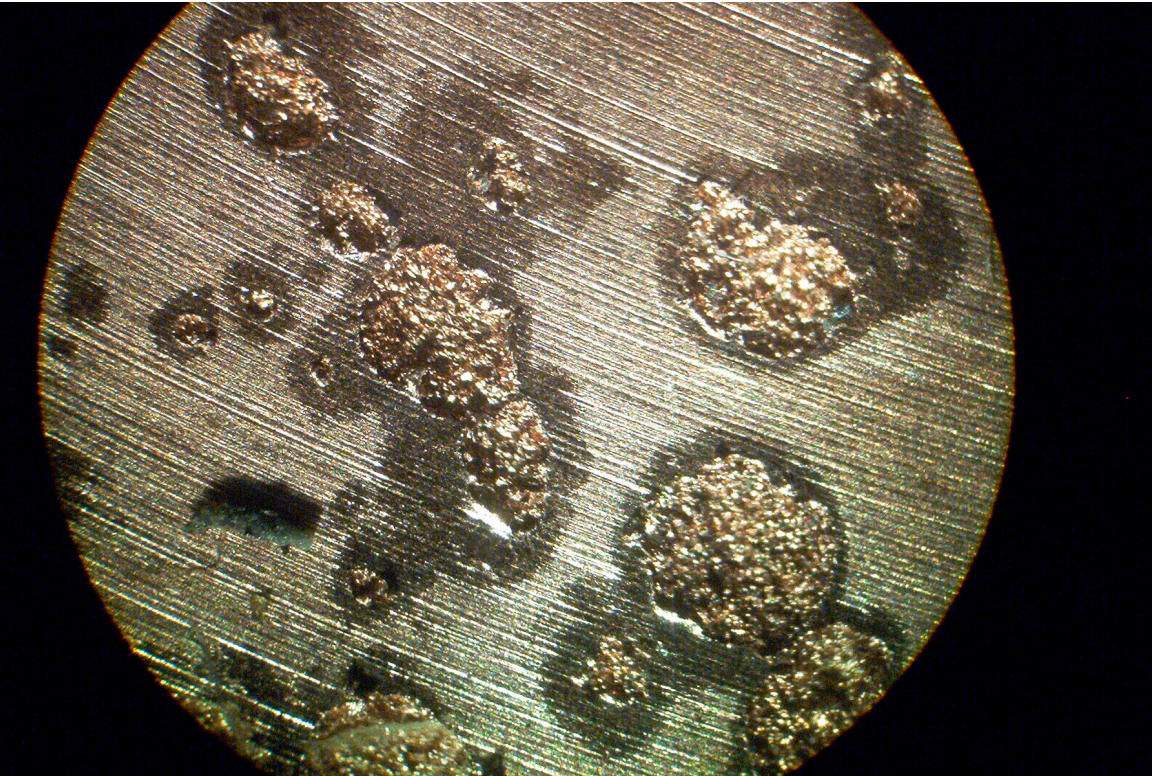
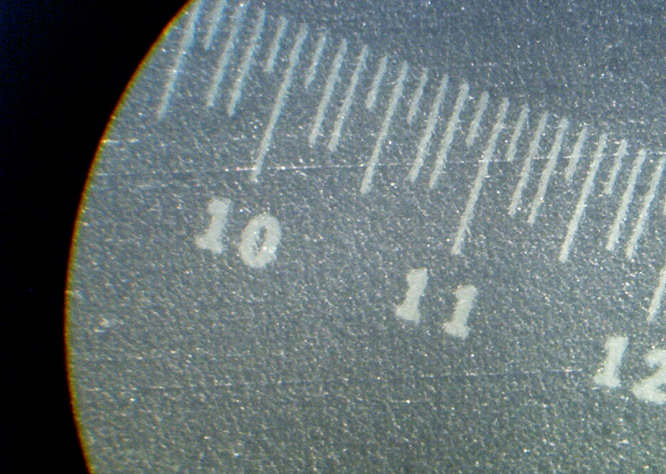
Atomic Layer  
 Deposition

Unipolar arc physics  
 Ablation mechanisms  
 Enhancement Spectra  
 Crack formation  
 Physical dim. of asperities  
 $E_{\text{max}}$  - damage equilibrium



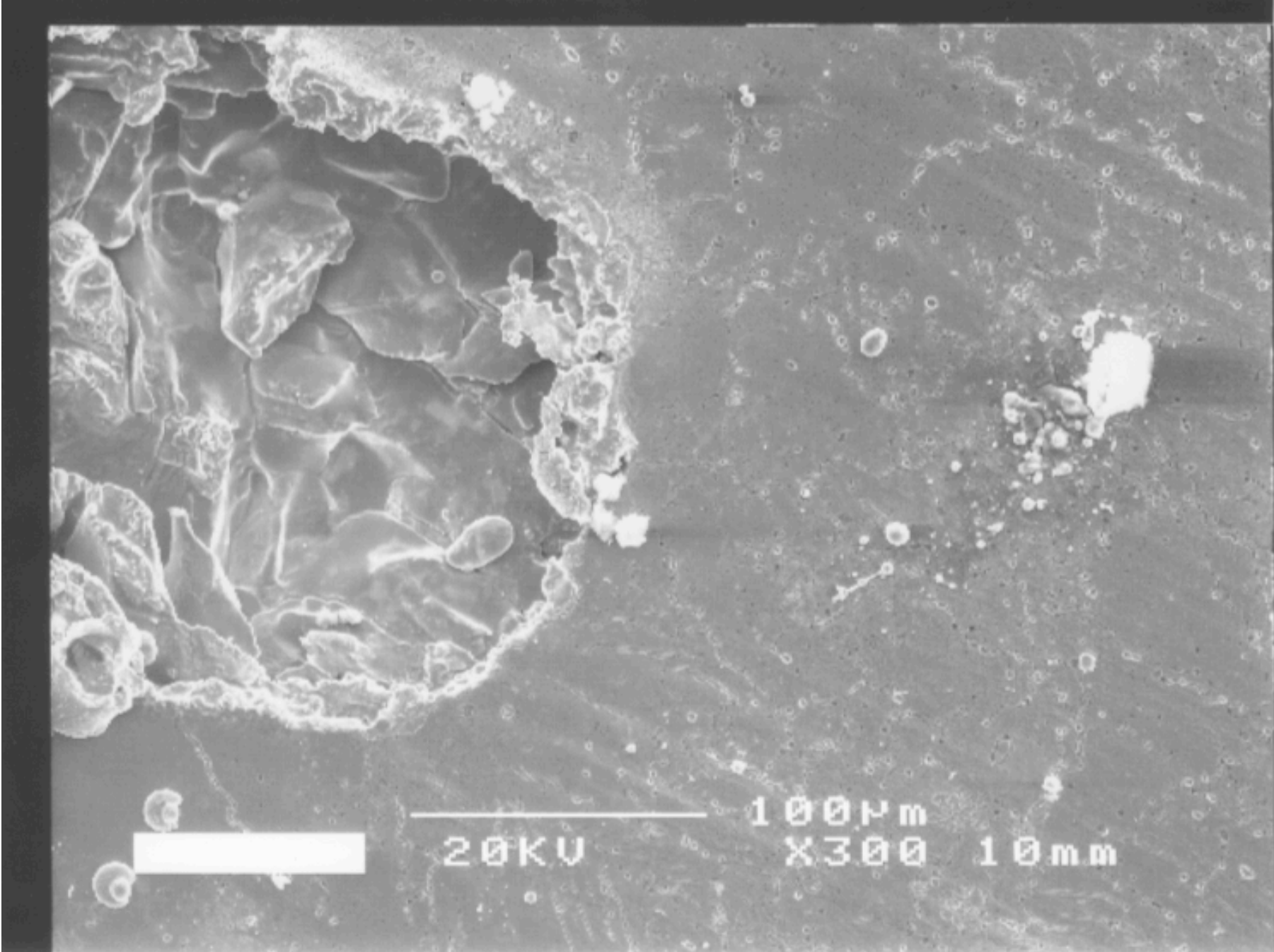


Microphotos of pits in the Cu plate from pillbox.

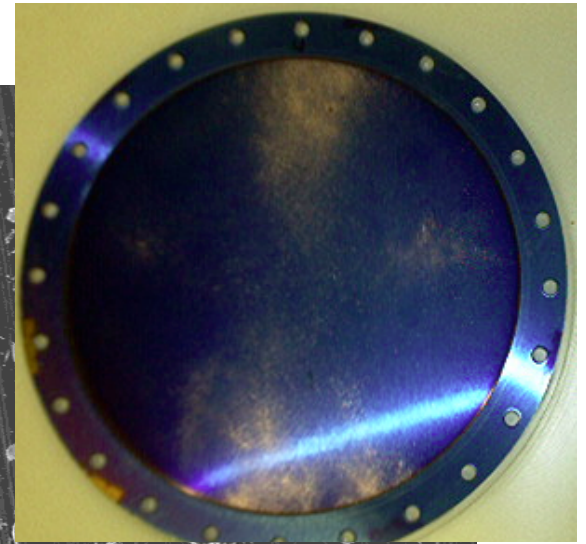
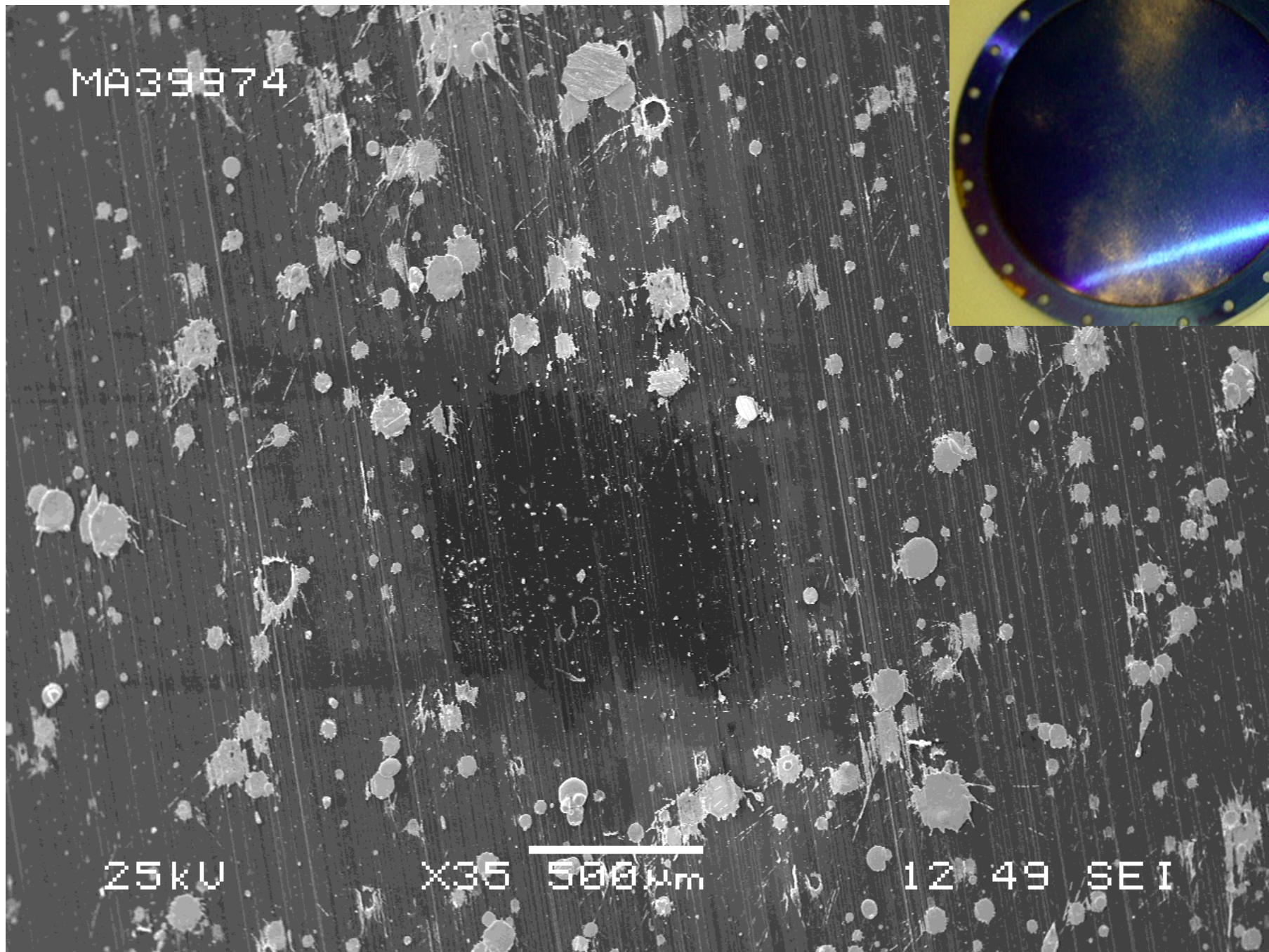




Surfaces are highly damaged.

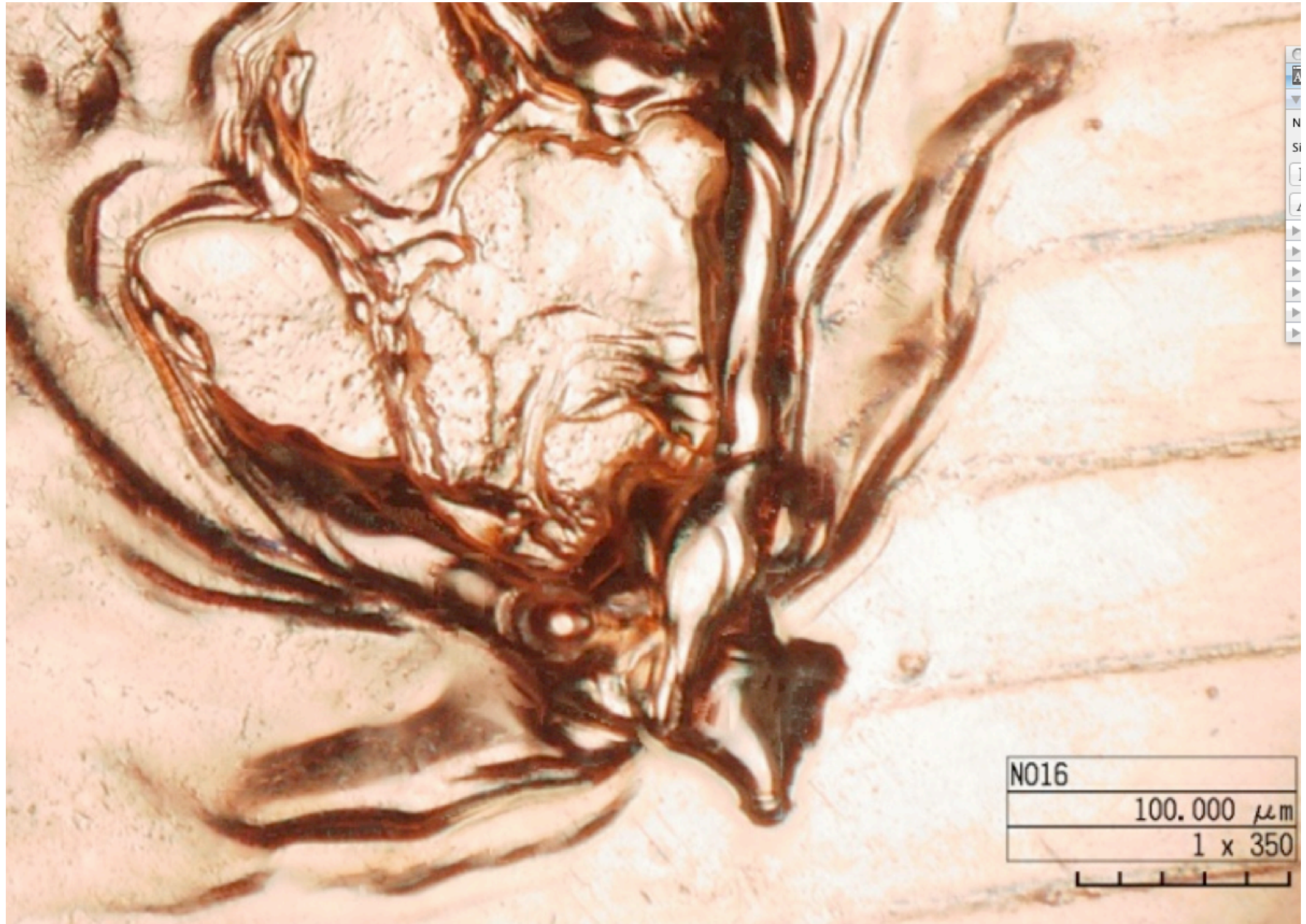


# McCrone SEM of Cu on Be.



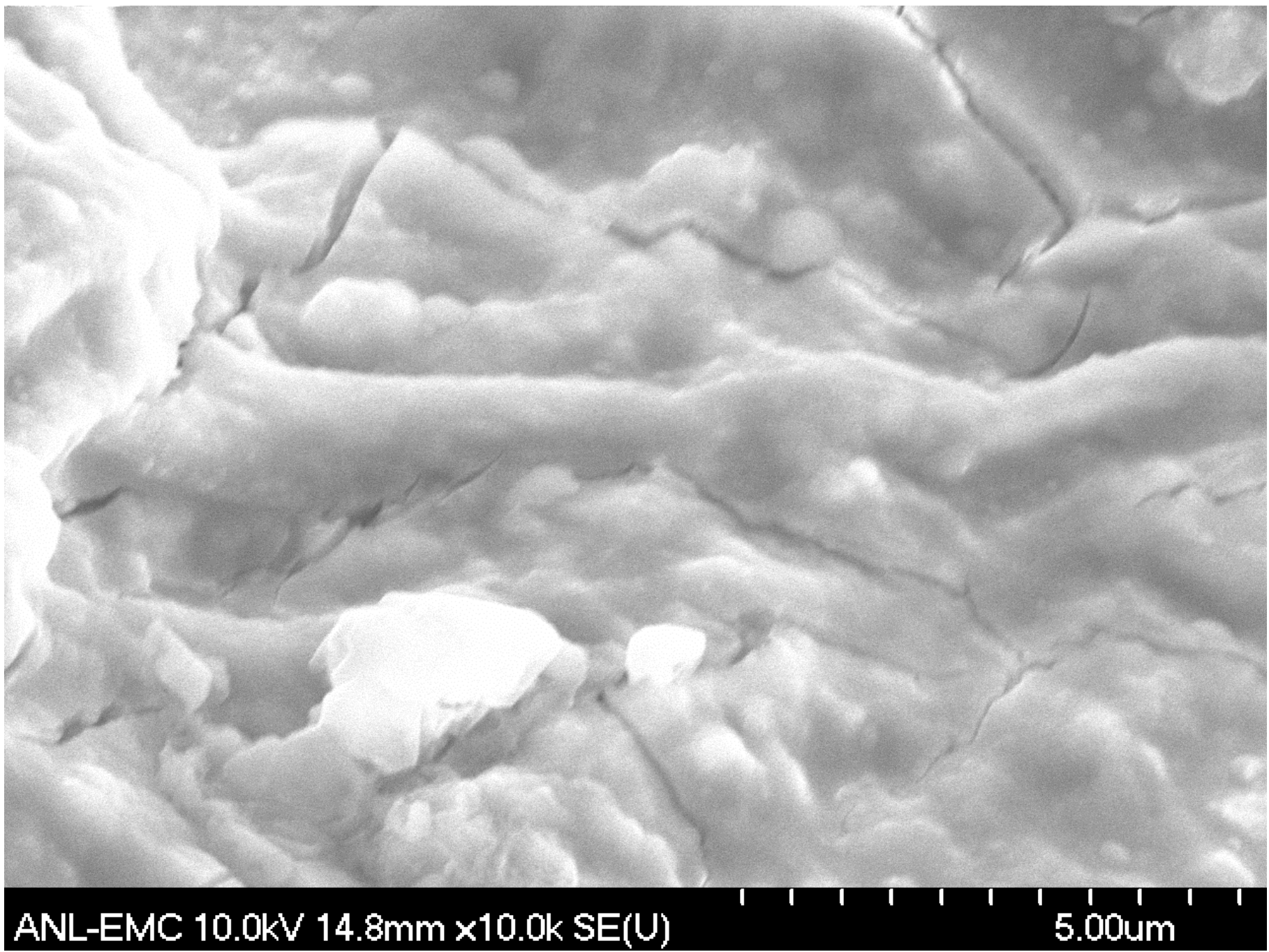


# Bob Rimmers multifocus camers.

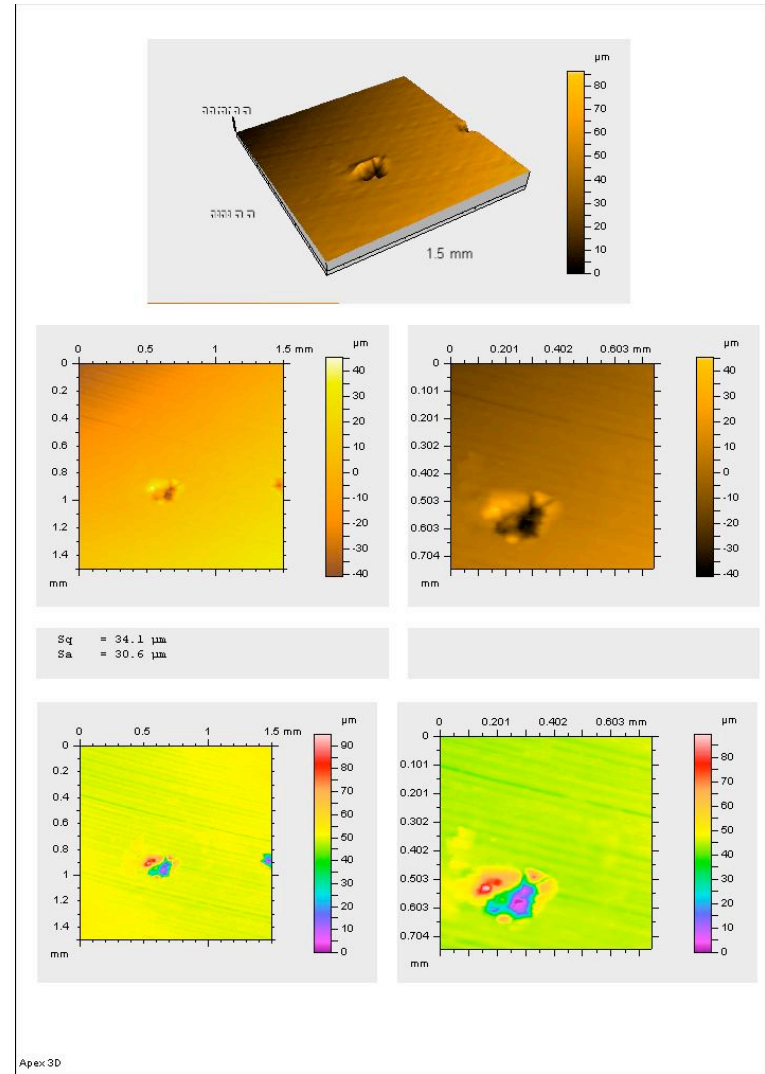
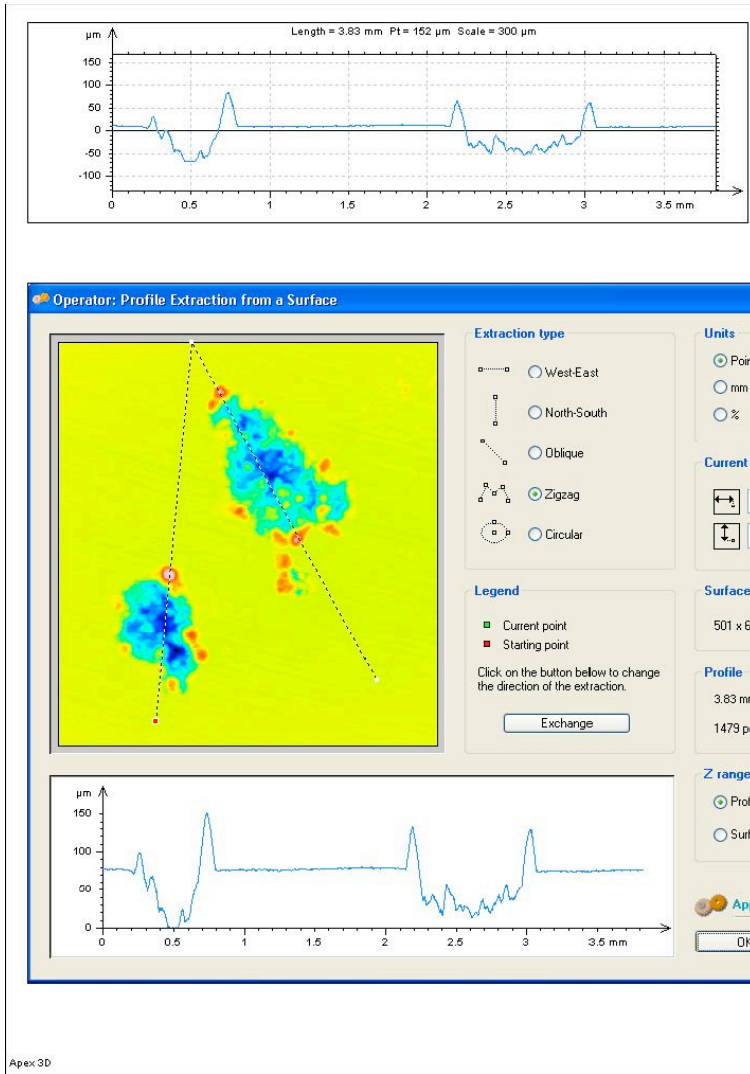




SEM picture of pit in copper plate.



# Profilometer measurements by Genfa Wu

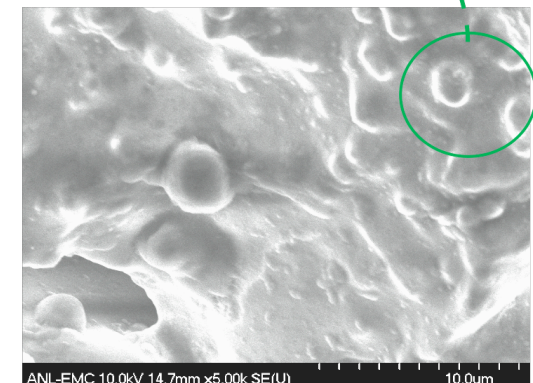
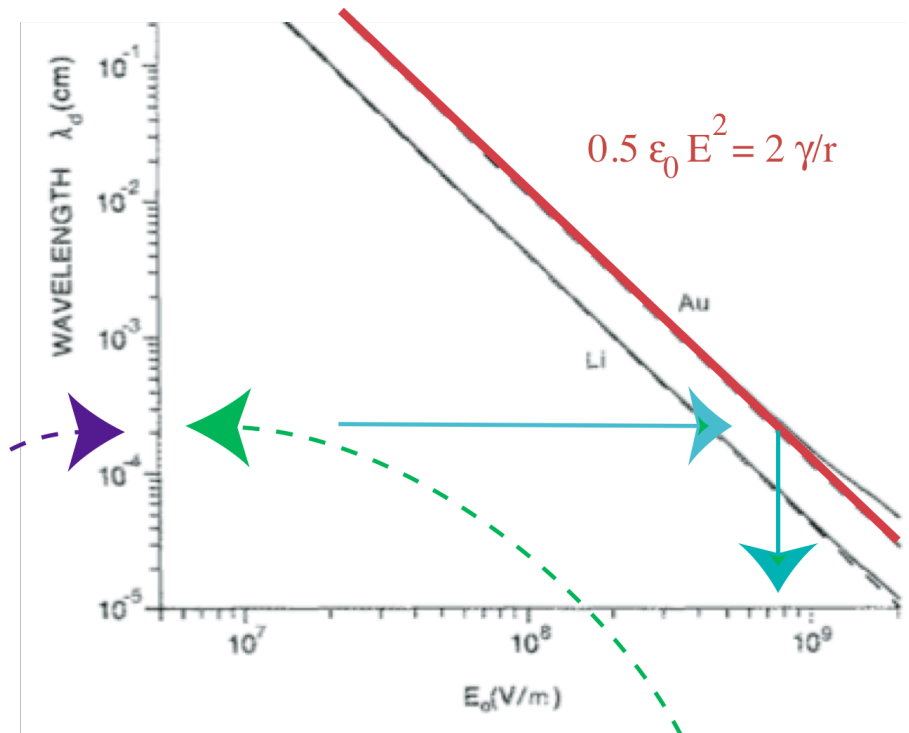
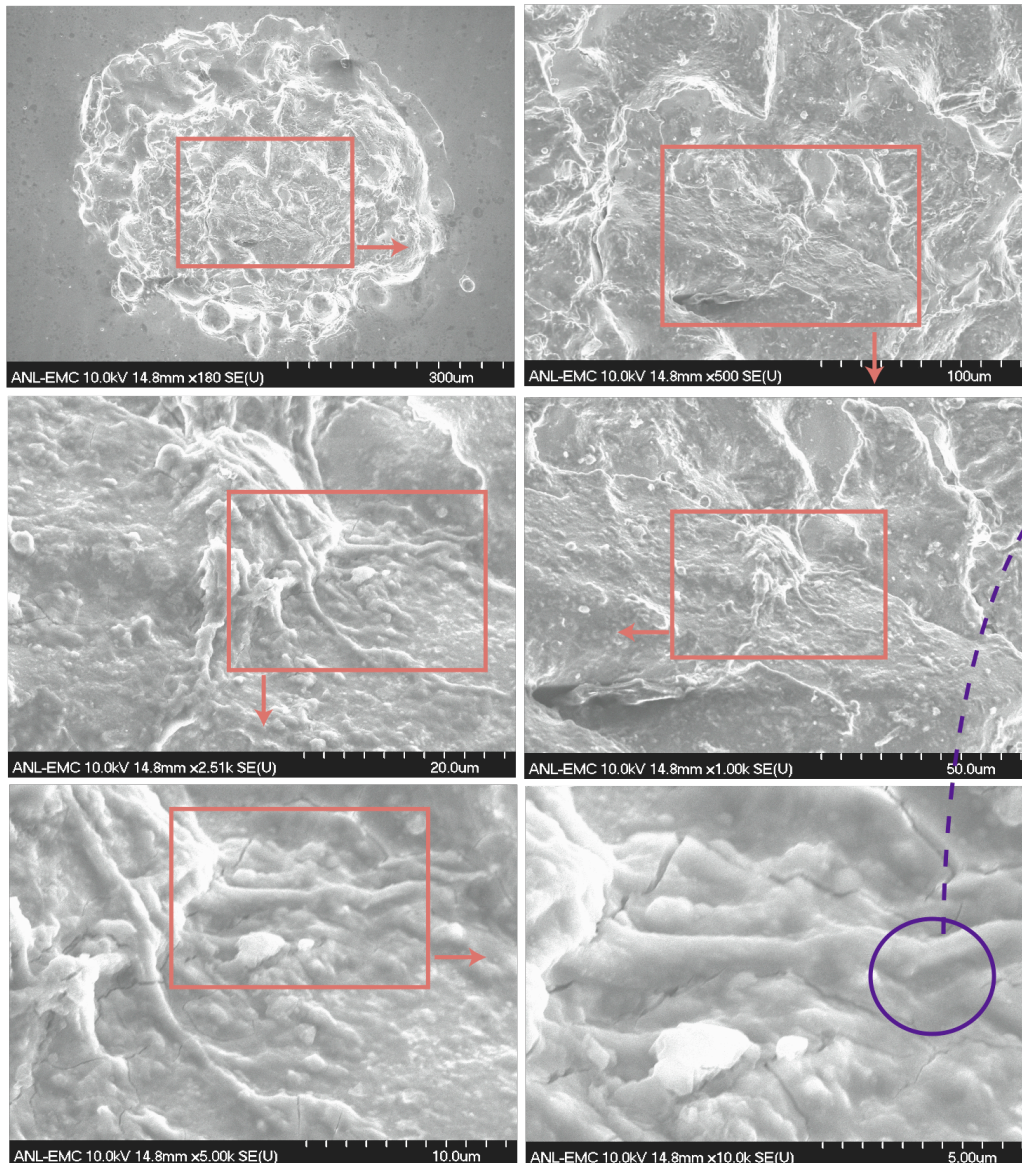




# The damage is not entirely random..

Things are chaotic at large dimensions, smoother at smaller ones, & structure  $\sim 2 \mu\text{m}$ .

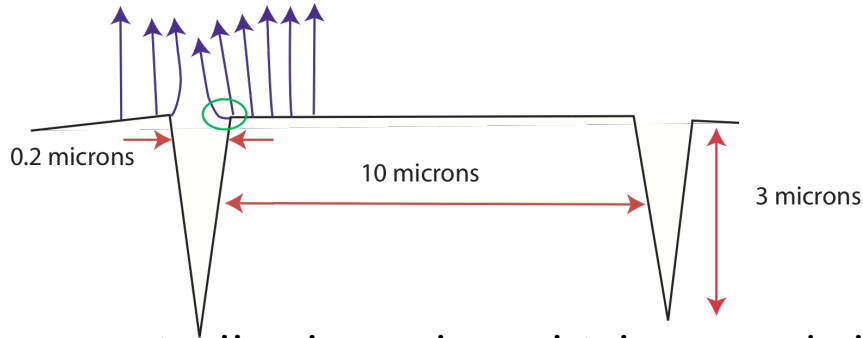
Setting electric pressure = surface tension gives an equilibrium.



# Cooling, cracks and $\beta$ 's:

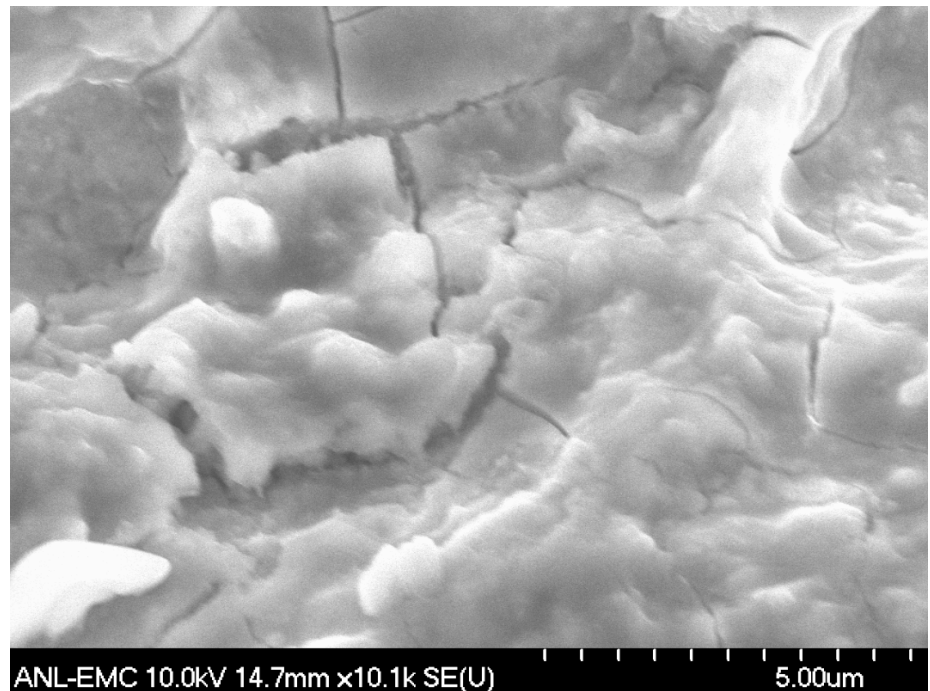
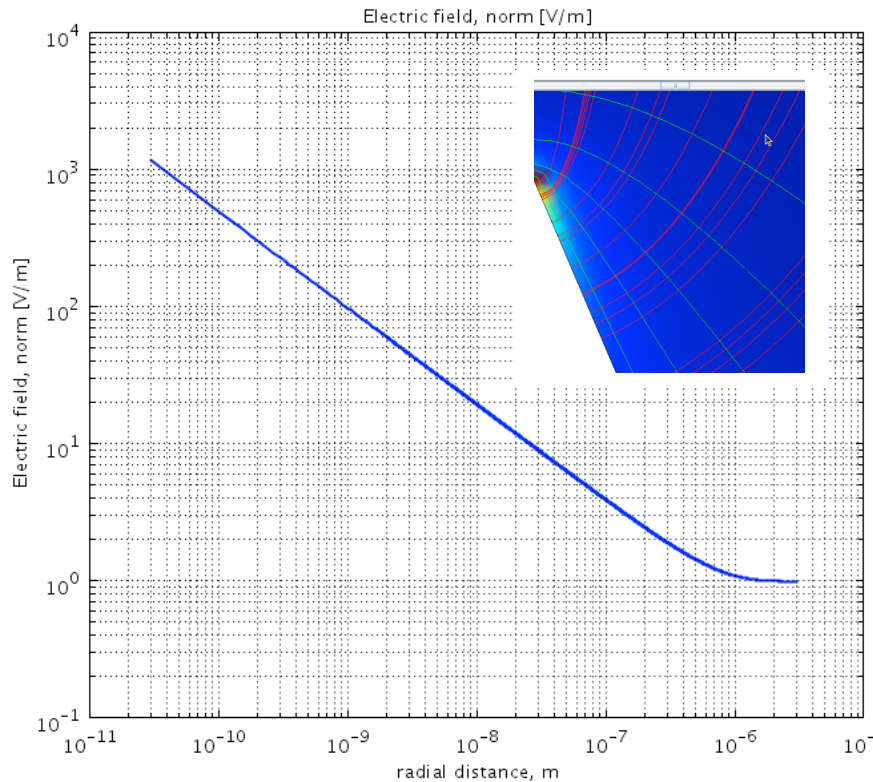
- Melted copper ( $\sim 3 \mu\text{m}$  thick, at  $\sim 1000 \text{ degC}$ ) cools and cracks.

Crack width:  $dx \sim (17 \times 10^{-6}) * 1000 * x \sim 2\% x, \quad x = 10 \mu \Rightarrow dx \sim 0.2 \mu.$



Can be modeled by a cone.

- Corners are atomically sharp, have high  $\beta$ s, and there are lots of them.



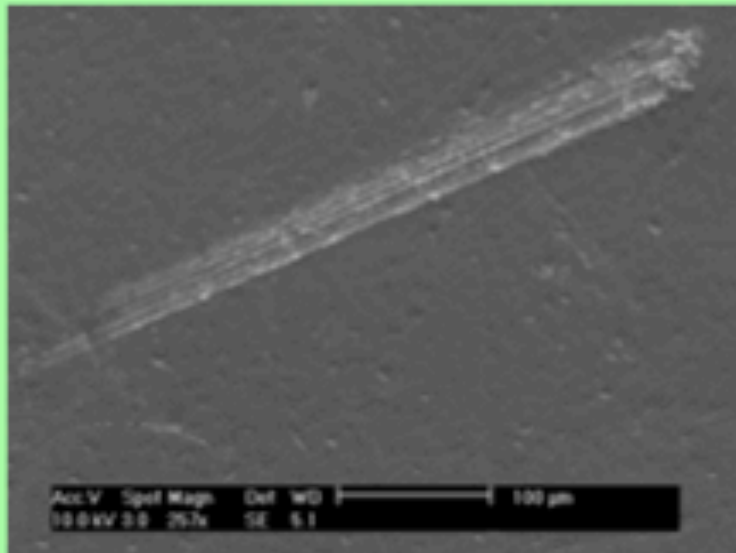
ANL-EMC 10.0kV 14.7mm x10.1k SE(U)

5.00um

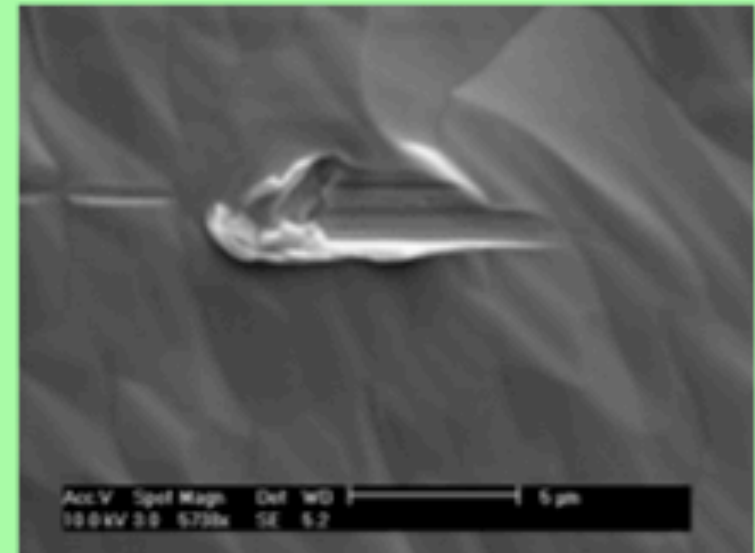


# Field emission microscope measurements show high $\beta$ s.

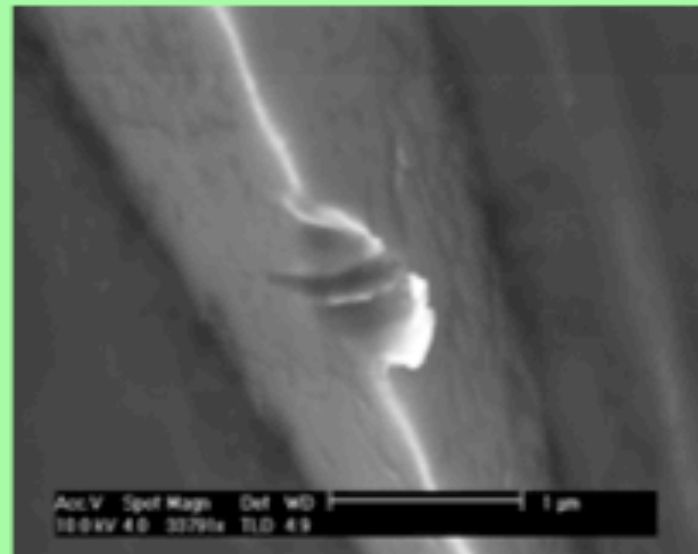
## Typical protrusion emitters containing only Nb (+ O?)



$E_{on}(2nA) < 60$  MV/m  
~500  $\mu\text{m}$  long scratch  
(mishandling of sample)



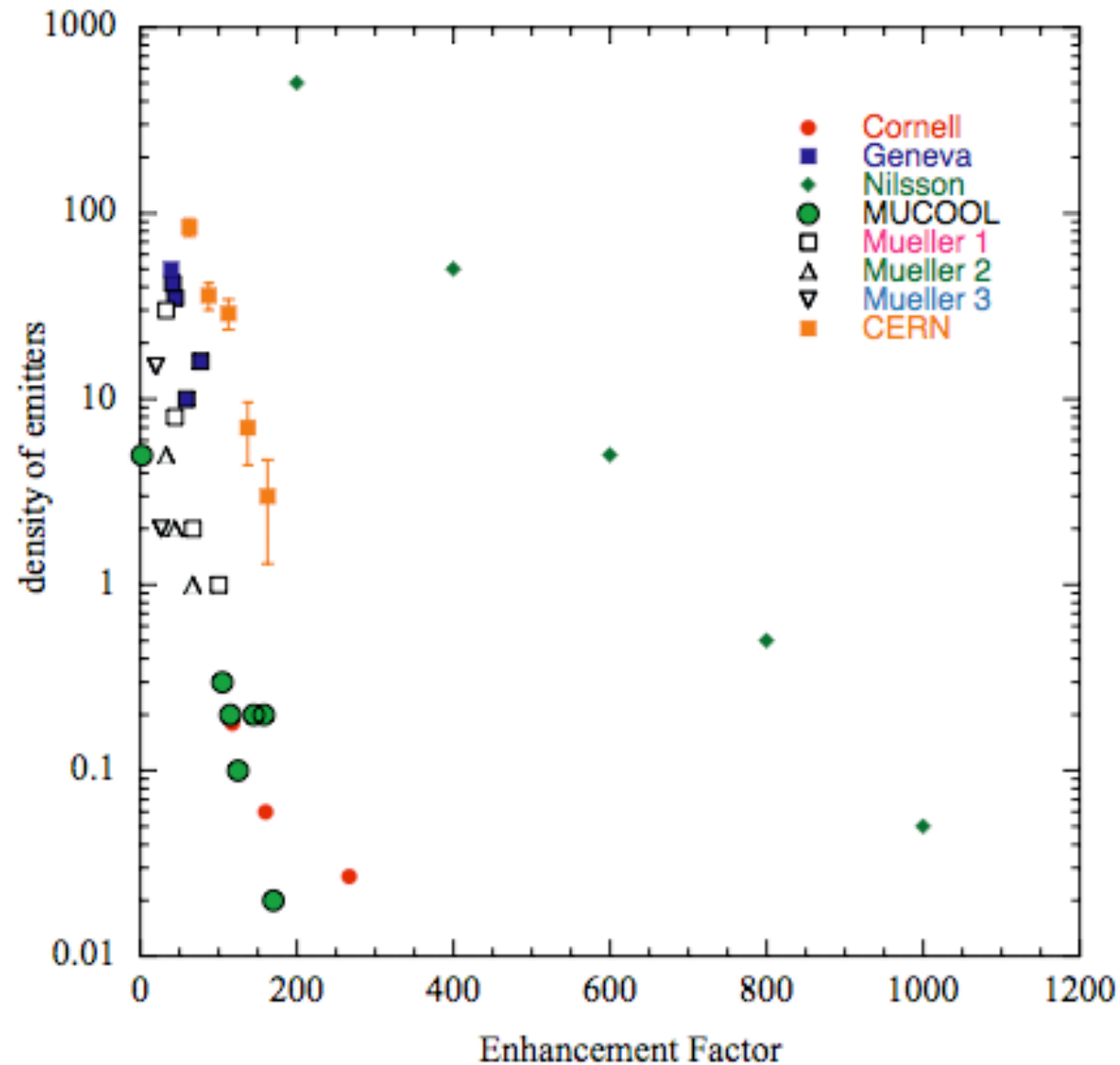
$E_{on}(2nA) = 90$  MV/m  
~5  $\mu\text{m}$  long groove  
 $\beta = 71$ ,  $S = 2.3 \cdot 10^{-6} \mu\text{m}^2$



$E_{on}(2nA) > 140$  MV/m  
~1  $\mu\text{m}$  small defect  
 $\beta = 59$ ,  $S = 7 \cdot 10^{-8} \mu\text{m}^2$

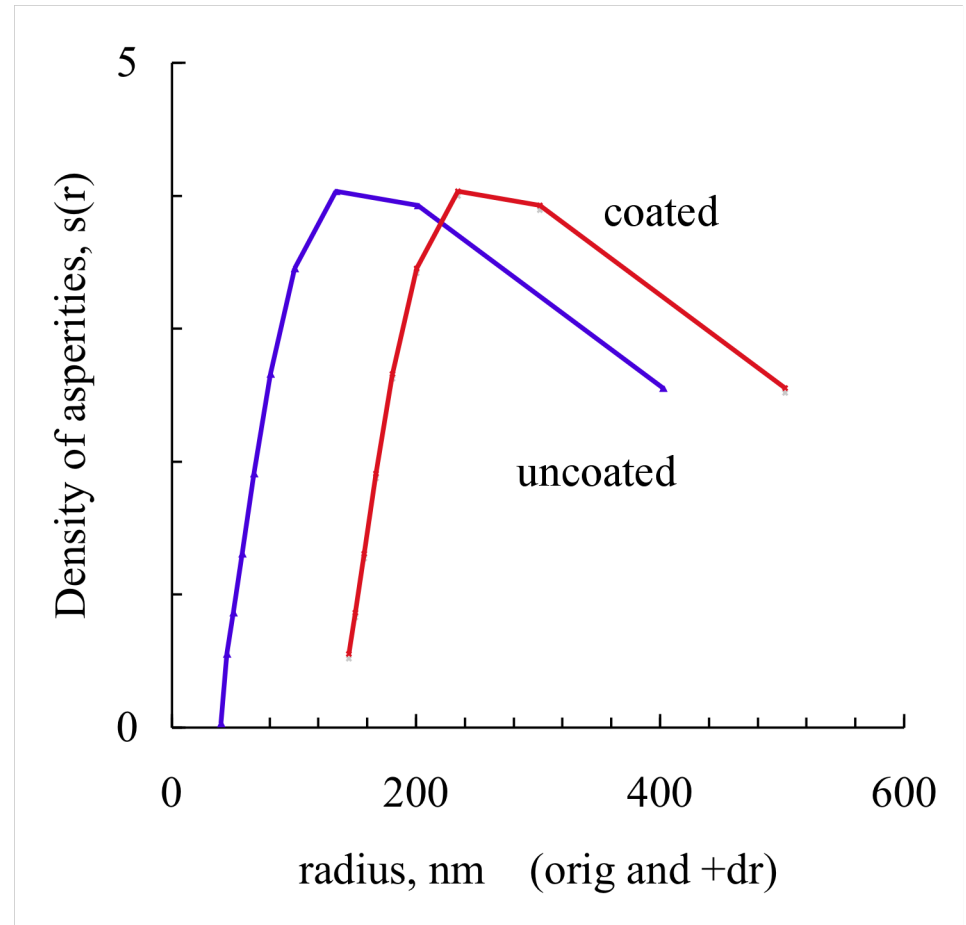
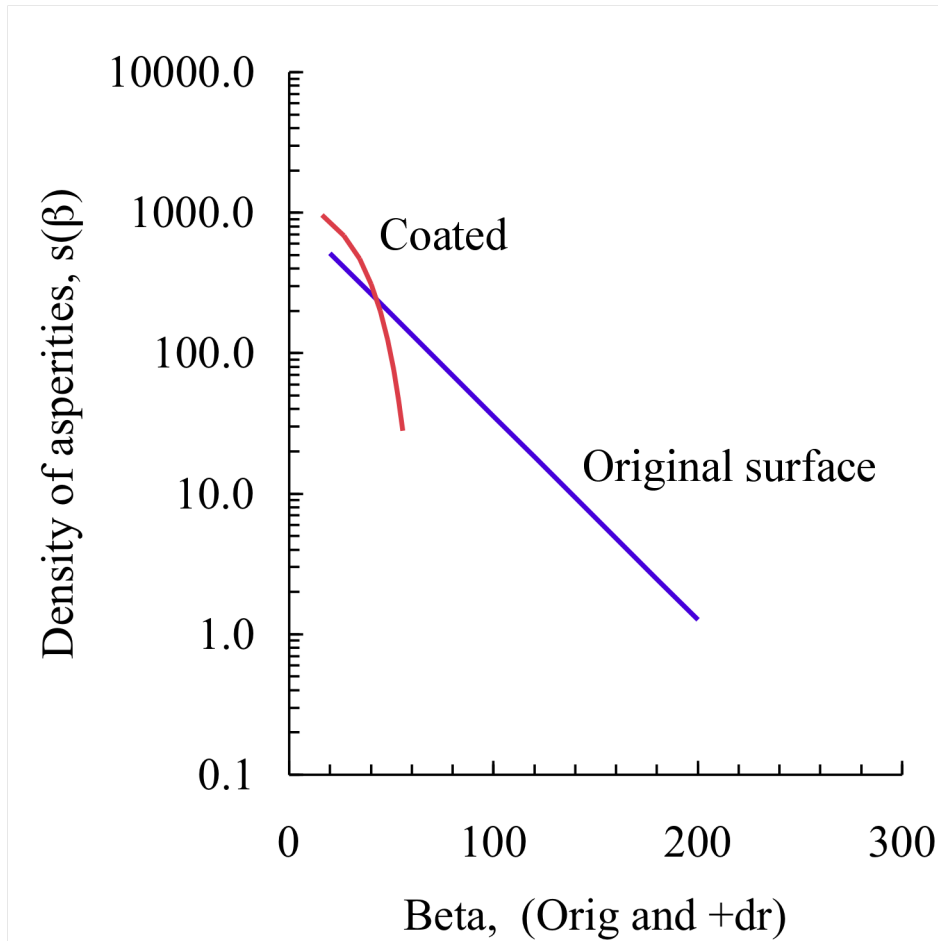
# There is a spectrum of enhancement factors.

- Everyone sees roughly the same thing.



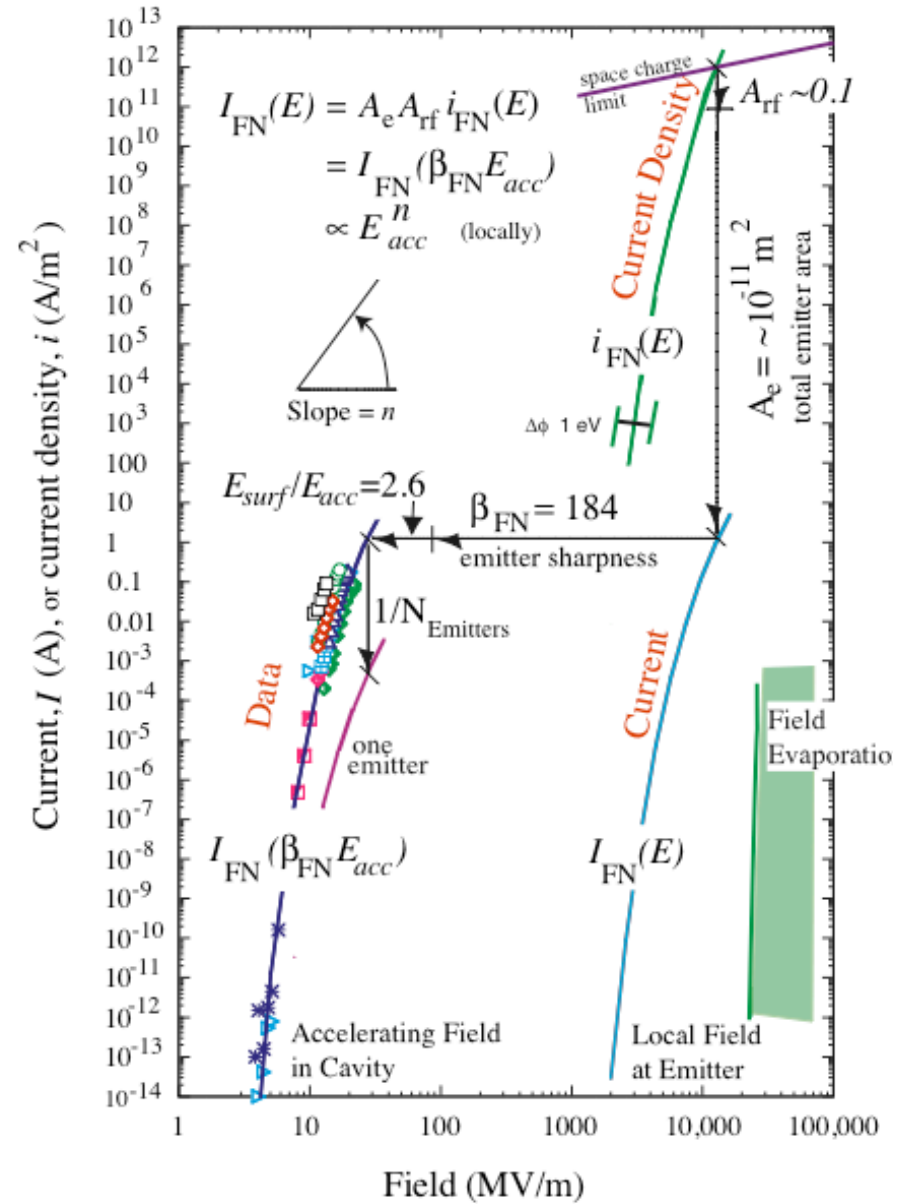
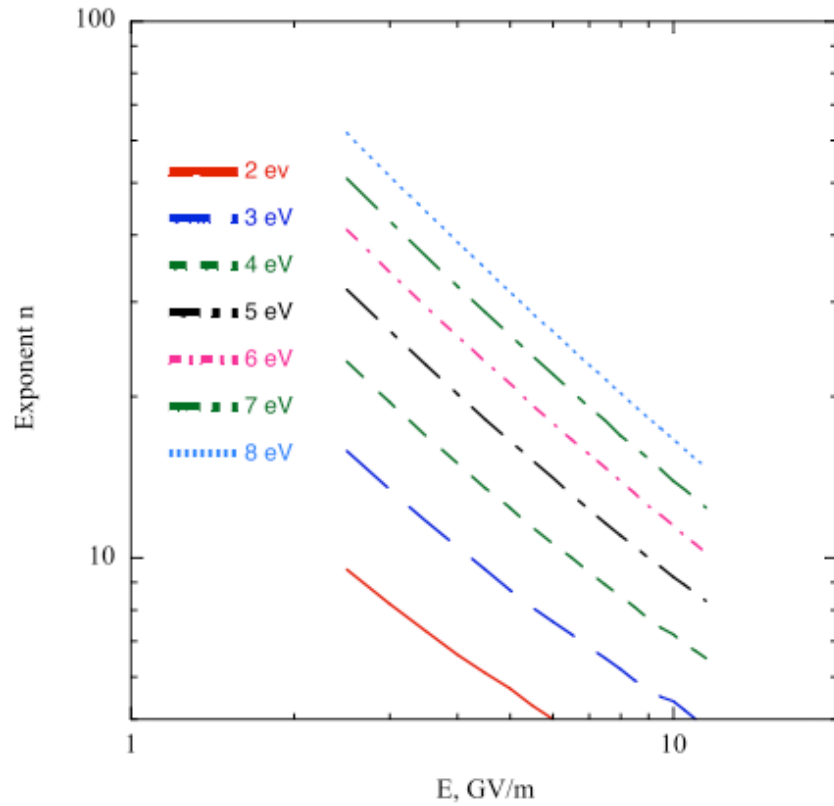
# Smoothing the surface should make cavities "Breakdown Proof".

- What is the effect of increasing the emitter radius?



- This argument is scale invariant.

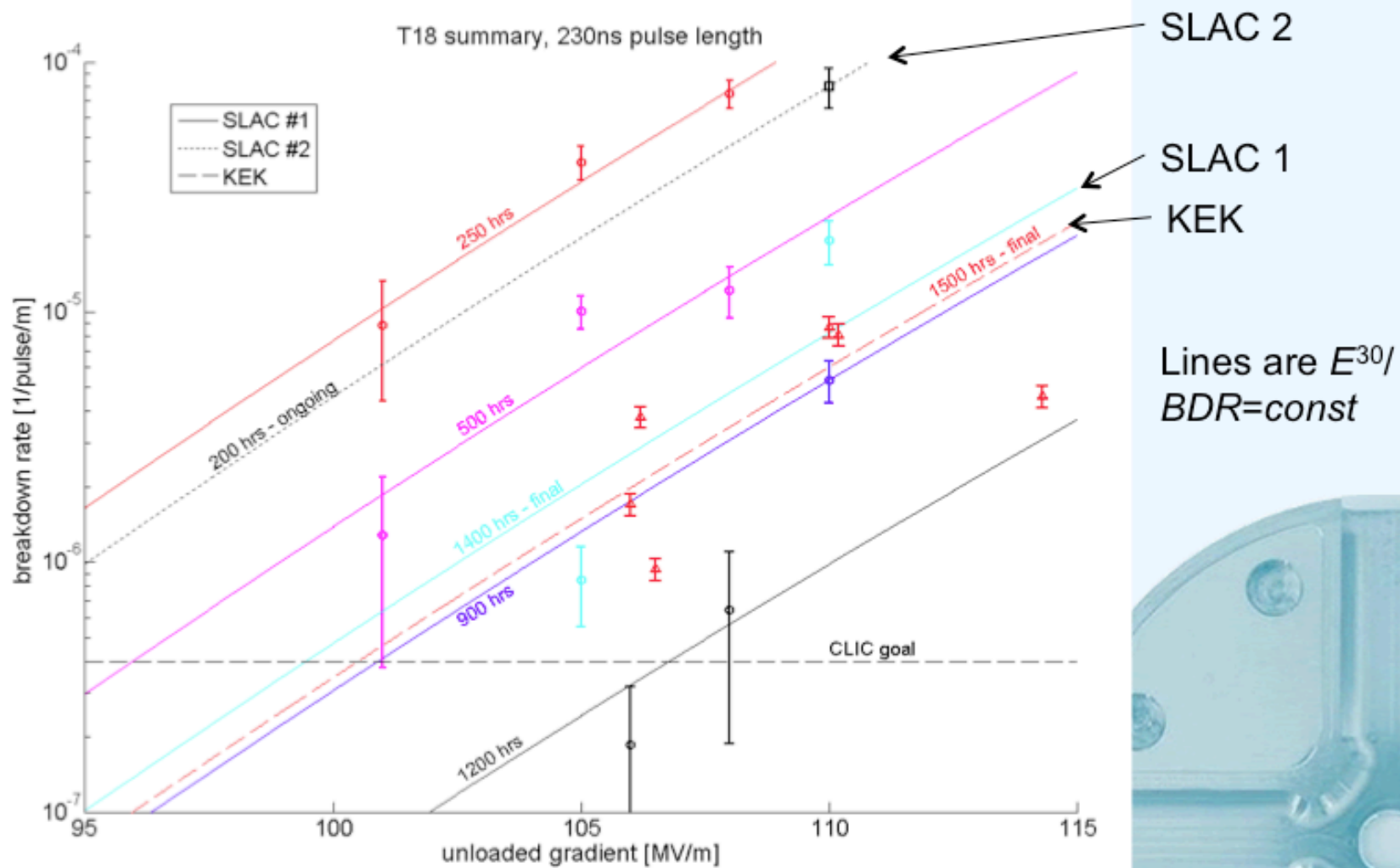
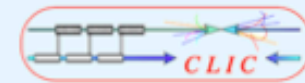
Field emission goes like  $E^{14}$ .



Breakdown rates go like  $E^{30}$ .



## CERN/KEK/SLAC T18 structure tests



## ALD can extend common sense to the nanoscale.

- We would never tolerate random cm or mm scale sharp points in cavities.
- We do have many asperities with radii  $\sim 4 - 10$  nm, which we cannot remove.
- Conformal coatings can eliminate small radii.
- Since  $I_{\text{FN}} \sim E^{14}$ , and  $\text{BDR} \sim E^{30}$ , and  $E \sim 1/r$ , increasing the radius of nano-asperities by a factor of three would reduce  $I_{\text{FN}}$  by  $\sim 10^7$  and the BDR by  $\sim 10^{14}$ .
- Atomic Layer Deposition can produce conformal, conducting coatings of a number of materials.
  - These coatings have been demonstrated at  $\sim 75$  MV/m in SRF structures.
  - The primary experimental problems are associated with power couplers etc.
  - Tungsten seems to be the best thing to try first.

## “Breakdown Proof”: proof-of-principle and RF cavity tests

- The primary questions are the radius of the asperities, and deposition chemistry.
- We are considering an experiment that uses pre-sharpened pins that we can sequentially coat and measure the field emission current. Depositing a known thickness of material conformally on the tips we should be able to “turn off” field emission, and measure the radius of the emitter (breakdown site).
- We can also do this in-situ in a cavity. The experiment would involve
  - First: Condition the cavity & measure the dark currents at maximum gradient.
  - Then: Coat the cavity with ALD using known thickness of different metals.
  - Measure the field emission as a function of ALD coating thickness.
- The cavity might look like this.

