# RADIATION EFFECTS ON FUSION MAGNET COMPONENTS

Harald W. Weber Vienna University of Technology Atominstitut, Vienna, Austria

Introduction: The ITER – Magnets Superconductors Stabilizer Insulation Conclusions

> ESS, 4<sup>th</sup> High Power Targetry Workshop, Malmö 5 May 2011





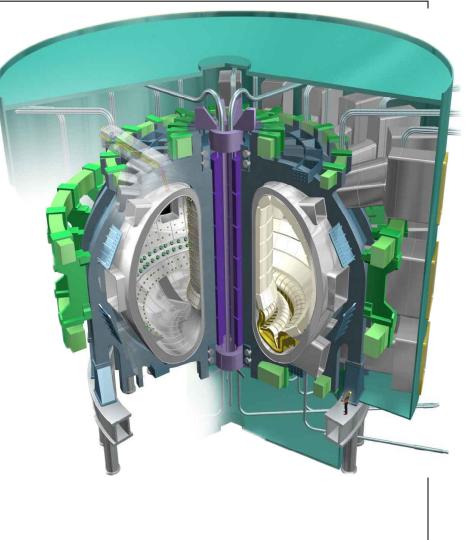


# INTRODUCTION

## **Overview: ITER**

#### **Main Parameters of ITER**

Total fusion power	500 MW
Q	≥ 10
Average 14Me∨ neutron wall loading	≥ 0.5 MW/m <sup>2</sup>
Plasma inductive burn time	300-500 s
Plasma major radius (R)	6.2 m
Plasma minor radius (a)	2.0 m
Plasma current (I <sub>p</sub> )	15 MA
Toroidal field at 6.2 m radius ( $B_T$ )	5.3 T

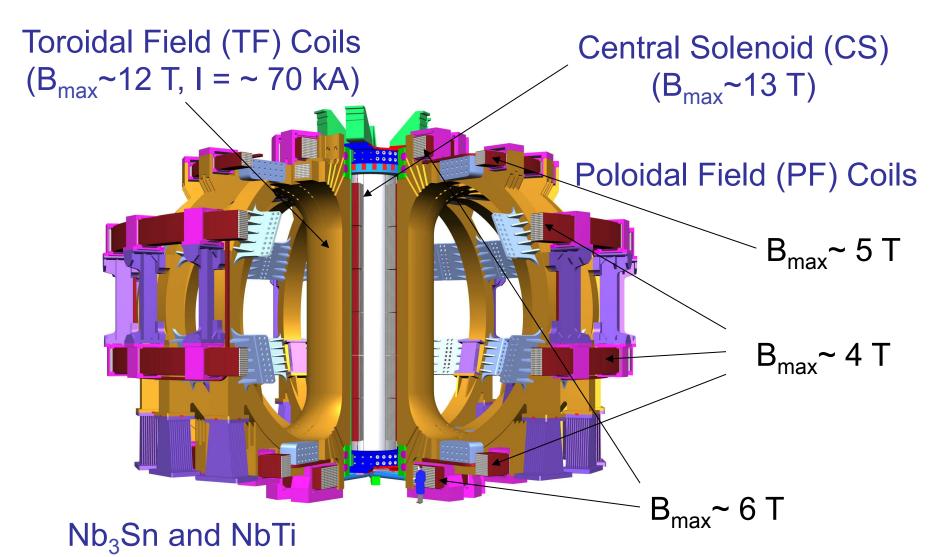








## ITER Magnet System (5 K / 6.5 K)









 The ITER project sets new limits for conductor and coil dimensions: Currents of up to 68 kA

Coils of up to 13 m (Nb<sub>3</sub>Sn) and 24 m (NbTi) in diameter

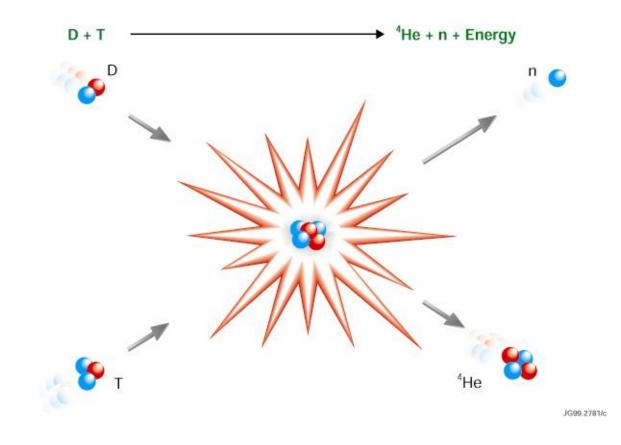
- More than 530 t of Nb<sub>3</sub>Sn strands are required for the TF and CS coils
- About 300 t of NbTi strands are required for the PF and CC coils
- HTS current leads are fabricated using Bi-2223 tapes up to 68 kA

# The ITER magnet system is a challenge for industry, worldwide ...









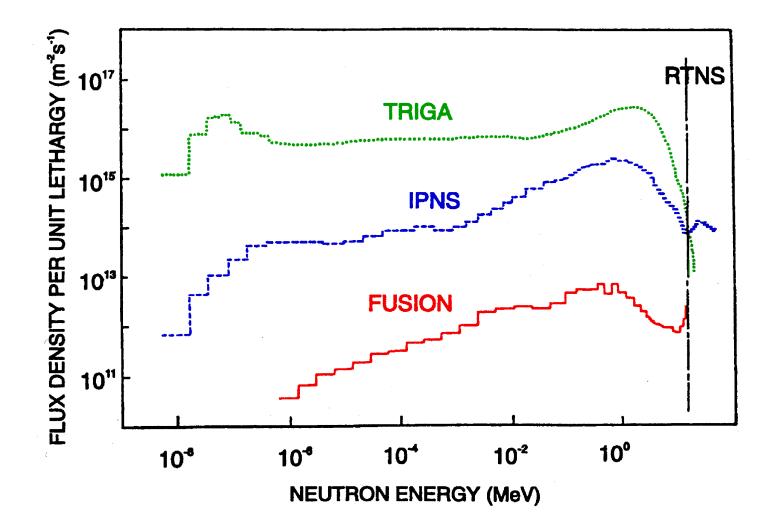
Production of 14 MeV neutrons – deposition of energy in the "first wall" → substantial materials problems (~1 MW/m<sup>2</sup>)!

At the magnet location: Attenuation by a factor of ~ 10<sup>6</sup>. Scattering processes lead to a "thermalization" of the neutrons!















## **DAMAGE ENERGY SCALING**

σ(E) T(E) F(E) t	neutron cross section primary recoil energy neutron flux density irradiation time in the	gy distribution
<	「(E) >	displacement energy cross section
E <sub>D</sub> = < σ(	(E) . T(E) > . F(E) . t	damage energy (total energy transferred to each atom in the material)

SUCCESSFUL SCALING OF  $T_c$  and  $J_c$  in metallic superconductors  $\Rightarrow$ 

PREDICTIONS OF PROPERTY CHANGES IN AN UNAVAILABLE NEUTRON SPECTRUM ARE FEASIBLE!



⇒





# **SUPERCONDUCTORS**

**Radiation will affect** 

### ➢ TRANSITION TEMPERATURE T<sub>c</sub>

- through disorder: @ unlikely in alloys

effective in metals and ordered compounds

### $\boxtimes$ NORMAL STATE RESISTIVITY $\rho_n$

- through the introduction of additional scattering centers

very small in alloys

significant in metals and ordered compounds

## ➢ UPPER CRITICAL FIELD H<sub>c2</sub>

- through the same mechanism:  $\rho_n \propto 1/I \propto \kappa \propto H_{c2}$ 

### I ⊂ CRITICAL CURRENT DENSITY J<sub>c</sub>

- through the production of pinning centers







## **DAMAGE PRODUCTION in LT SUPERCONDUCTORS**

#### FAST NEUTRONS (E > 0.1 MeV)

Displacement cascade initiated by the primary knock-on atom, if its energy exceeds 1 keV

#### **EPITHERMAL NEUTRONS (1 – 100 keV)**

Point defect clusters

#### THERMAL NEUTRONS

Transmutations, point defects

γ-rays: No influence

NB: Stable collision cascades in materials with low conductivity, e.g. HTS







## RESULTS

The "Workhorse": NbTi

A15 Superconductors:

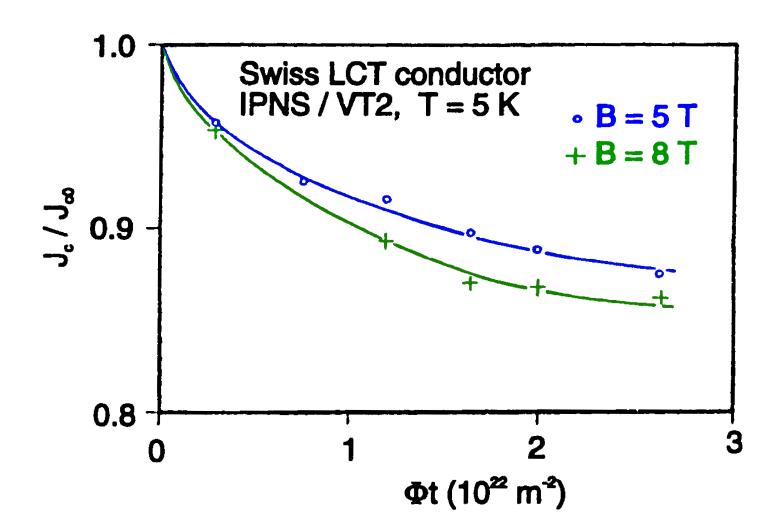
• Nb<sub>3</sub>Sn

- Alloyed A15' s: (Nb,Ti/Ta)<sub>3</sub>Sn
- Advanced A15' s: Nb<sub>3</sub>Al















**Results on NbTi** 

SMALL EFFECTS on  $J_{\rm c}\,$  - depending on the initial micro-structure for flux pinning

SMALL DECREASE of  $\rm H_{c2}$  - caused by a

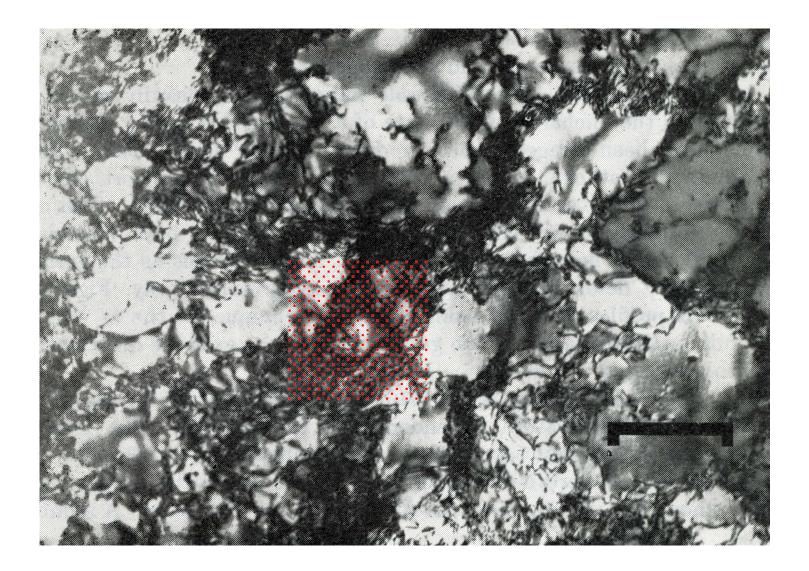
SMALL DECREASE of  $\rm T_{c}$ 

- Results typical for materials with a *high degree of disorder*
- Initial optimized defect structure for flux pinning is "disturbed"





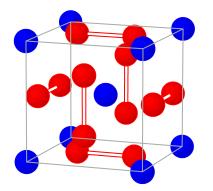




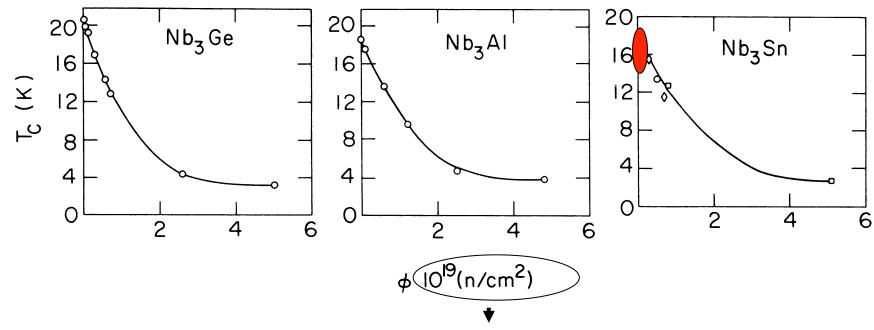








## **A15 SUPERCONDUCTORS**

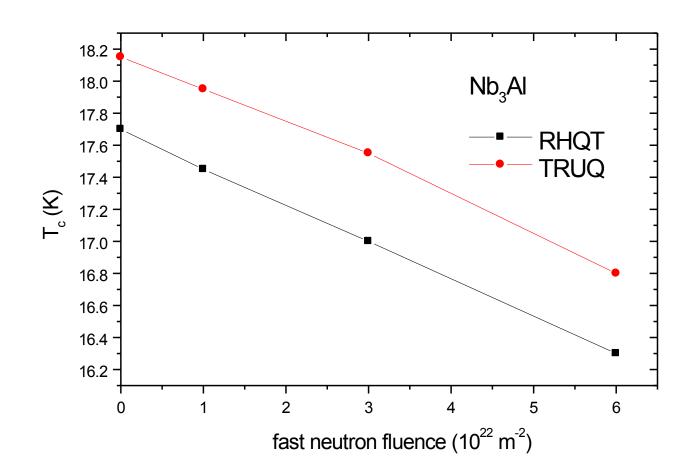


!! Scale not accurate: maximum fluence around 7-10 x 10<sup>23</sup> m<sup>-2</sup> !!





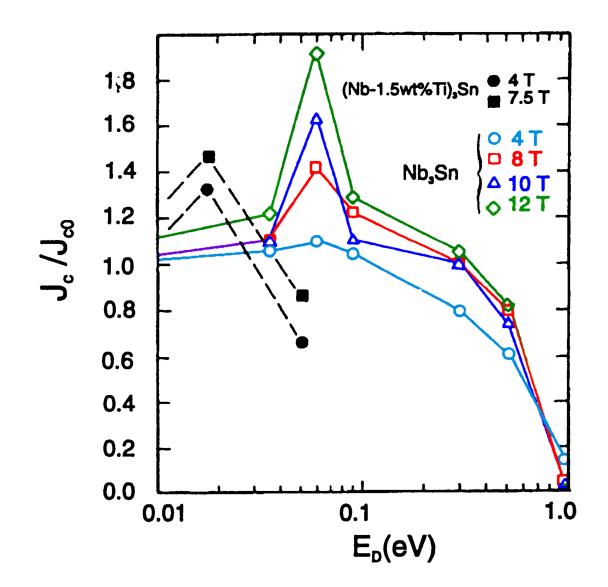








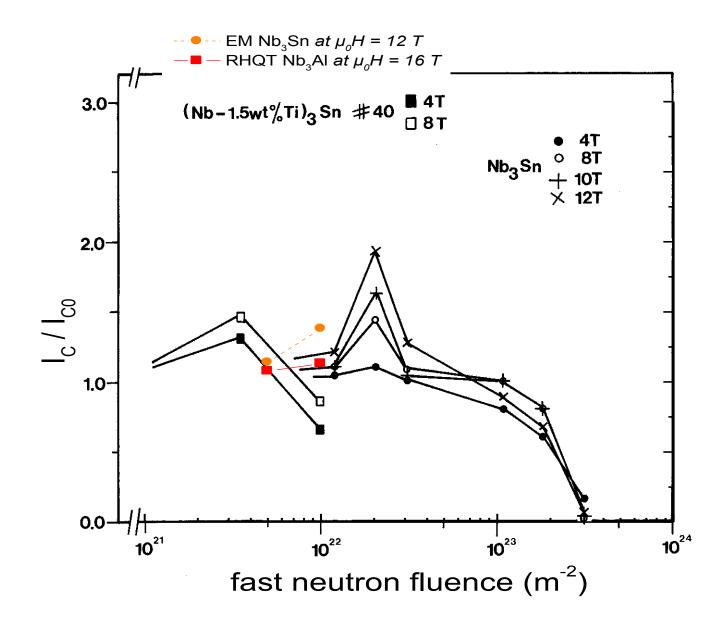










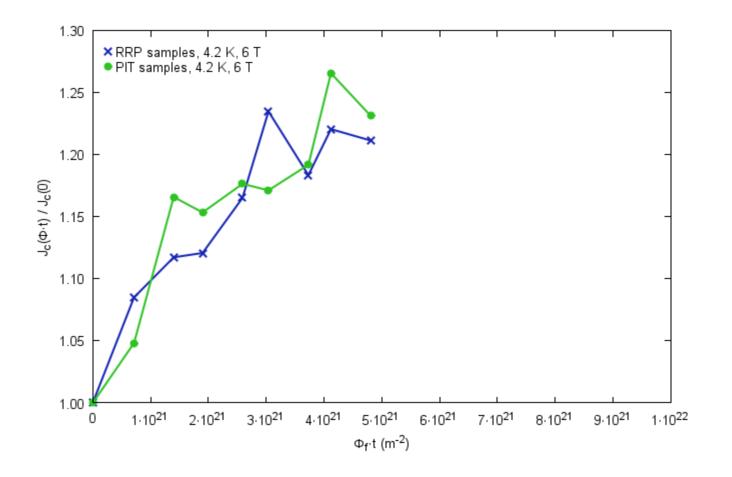








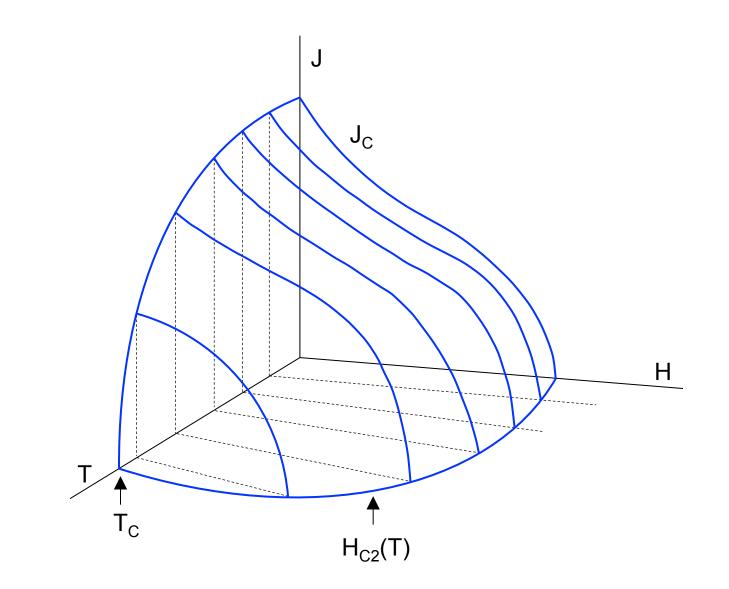
#### RRP (OST): (NbTa)<sub>3</sub>Sn – PIT (Bruker EAS): (NbTa)<sub>3</sub>Sn







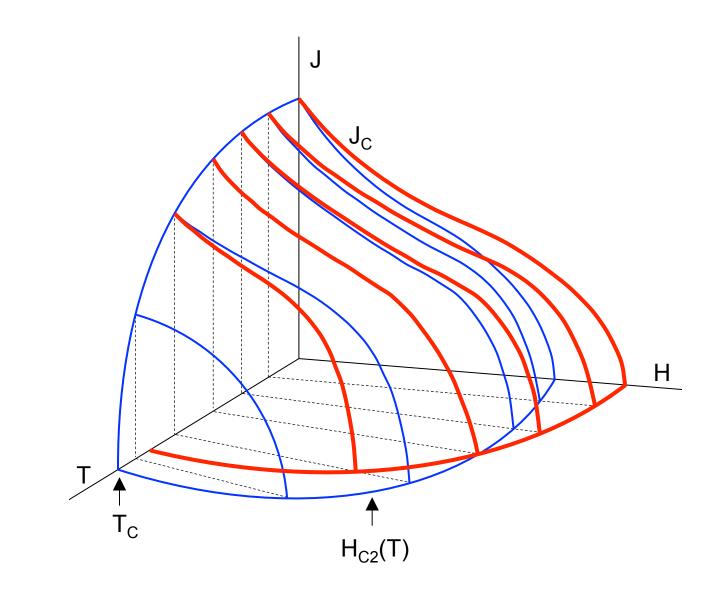


















## SUMMARY: Nb<sub>3</sub>Sn

SIGNIFICANT (and later on drastic) EFFECTS on  $T_c$  - caused by disorder SIGNIFICANT ENHANCEMENTS OF  $J_c$  (followed by a precipitous drop)

increase caused by an increase of U mean free noth offect

- increase caused by an increase of H<sub>c2</sub> mean-free-path effect
- drop caused by the  $\rm T_{\rm c}$  degradation

Typical for materials with *a high degree of order* 

## **SUMMARY: alloyed Nb<sub>3</sub>Sn** (Addition of small amounts of Ti or Ta)

Mean-free-path effect increases  $H_{c2} \Rightarrow ENHANCEMENT \text{ OF } J_c$ 

**But** additional scattering centres due to neutron irradiation lead to an *earlier* decrease of  $J_c$  (at lower fluence)

## Similar results on Nb<sub>3</sub>Al









Normal state resistivity essential for stabilization and quench protection

In-field resistivity experiments on copper

Irradiation *must* be done at low temperature (~ 5 K) due to substantial annealing

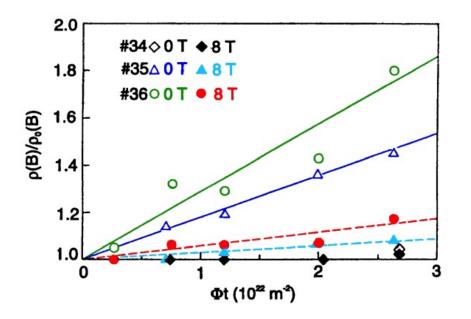
(most low temperature irradiation facilities have been shut down, only one 14 MeV source available in Japan)







- Resistivity measurement at 10 K
- Neutron irradiation at the IPNS spallation source at 5 K
- Warm-up cycle to RT
- Resistivity measurement at 10 K



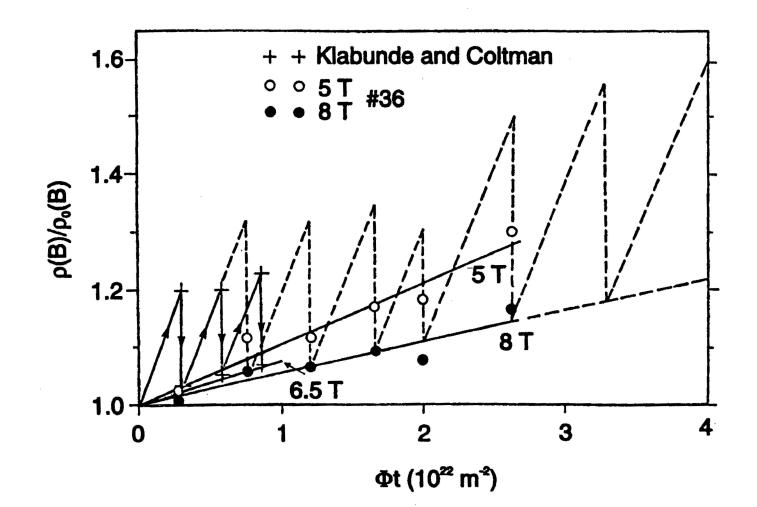
Multifilamentary NbTi-conductors

#34: RRR ~ 60 #35: RRR ~ 120 #36: RRR ~ 120

















Most critical component of the magnet in a radiation environment

Has to provide electrical insulation ( $\checkmark$ )

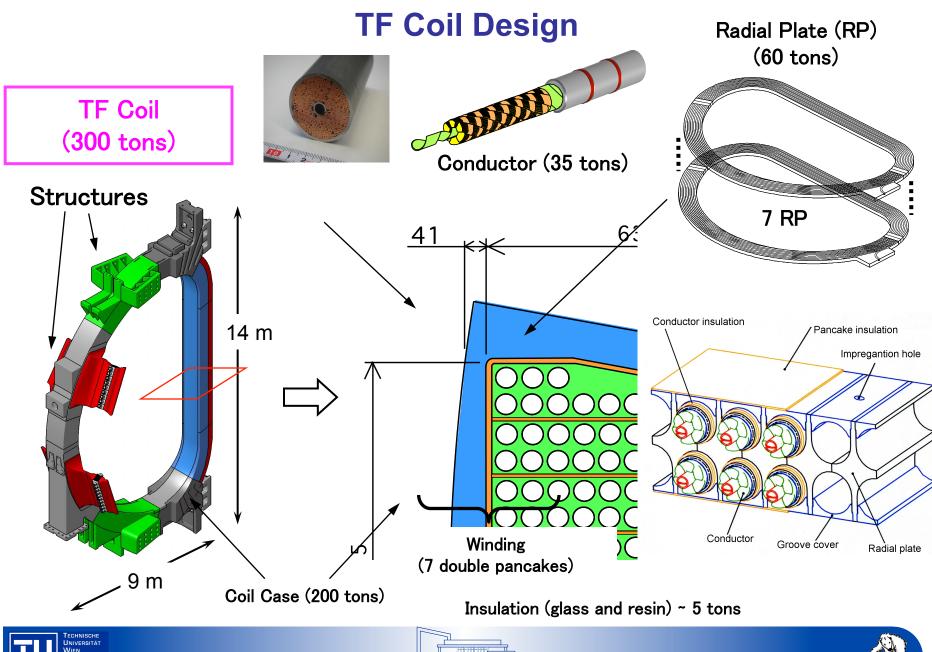
Has to provide **mechanical strength** and to withstand thermal contraction / expansion and Lorentz forces

Must be suitable for a vacuum-pressure impregnation process – "pot life"









VIENNA



LOW TEMPERATURE PHYSICS



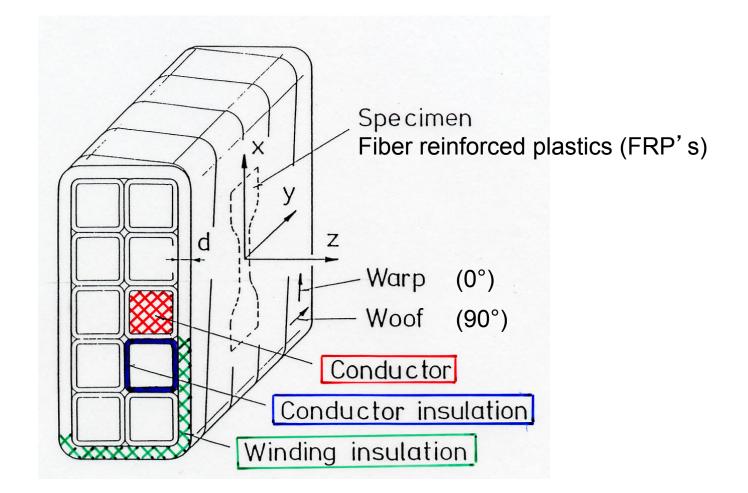








## **Typical magnet insulation build-up**

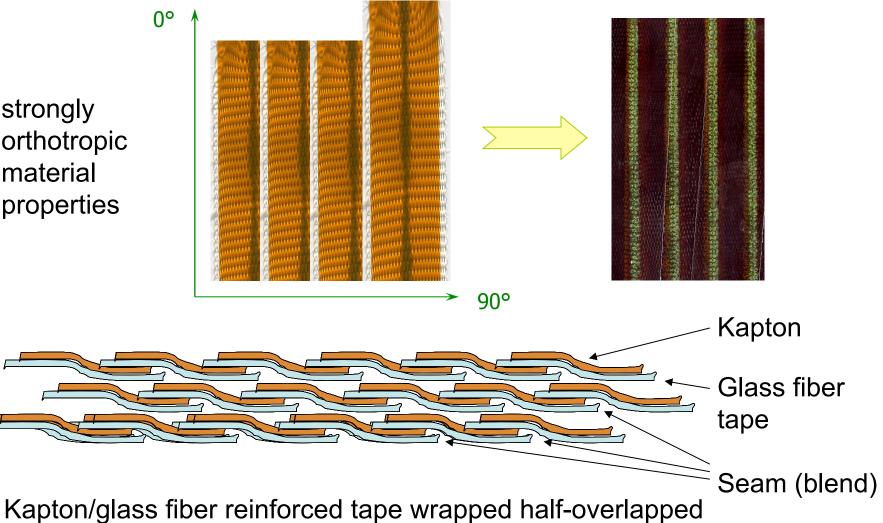








strongly orthotropic material properties



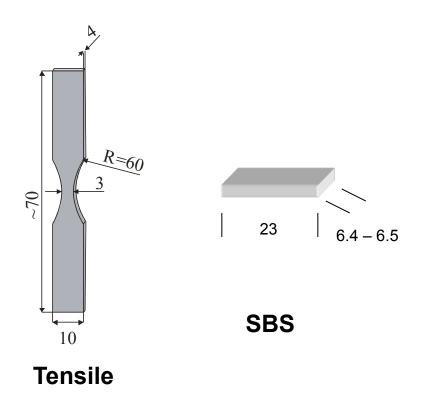
Technische Universität UNIVERSITY OF TECHNOLOGY





# **Test procedures**

#### Test specimen



All tests @ 77 K

Static and dynamic tensile tests (90 ° direction)

Short-beam-shear (SBS) test with span to thickness ratio of 4:1 and 5:1 (0 ° and 90 ° direction)

Neutron irradiation in the TRIGA reactor (Vienna) to a fast neutron fluence of 1, 2 and  $4x10^{22}$  m<sup>-2</sup> (E > 0.1 MeV)

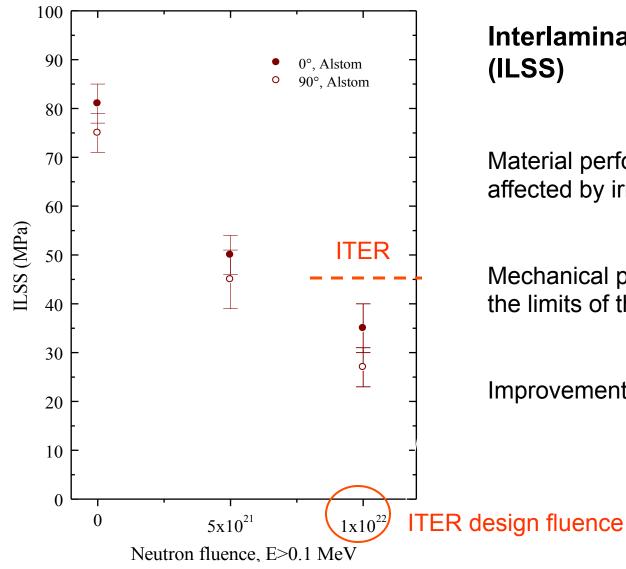
Total absorbed dose of ~50, 100 and 200 MGy







#### **EUHT DGEBA Epoxy (Alstom)**



# Interlaminar shear strength (ILSS)

Material performance drastically affected by irradiation

Mechanical properties are close to the limits of the ITER specifications

Improvement of the matrix stability!!







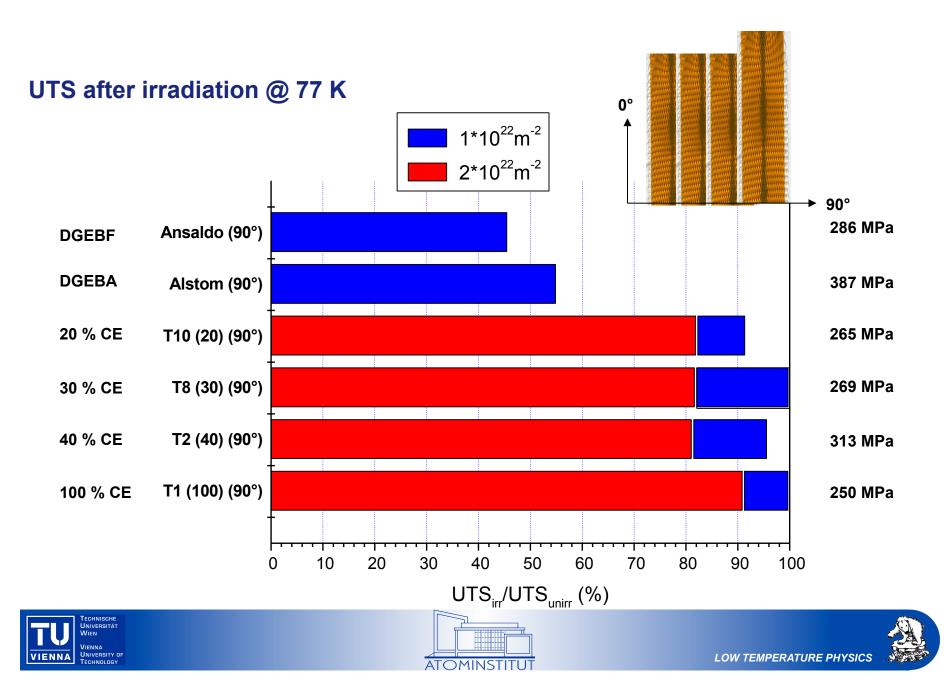
## **CE / epoxy blend**

	AroCy L-10	PY 306
Safety precautions	Avoid local overheating (hot spots) Store in sealed containers in dry rooms Provide sufficient air exchange Take necessary actions to avoid static electricity	Provide sufficient air exchange Take necessary actions to avoid static electricity Avoid strong acids and bases
Viscosity	η <sub>25 °C</sub> = 120 mPa s η <sub>60 °C</sub> = 17 mPa s	η <sub>25 °C</sub> = 1200-1600 mPa s
Pot life at high quantities	Dependent upon type and concentration of co-catalyst and catalyst used	Can be handled

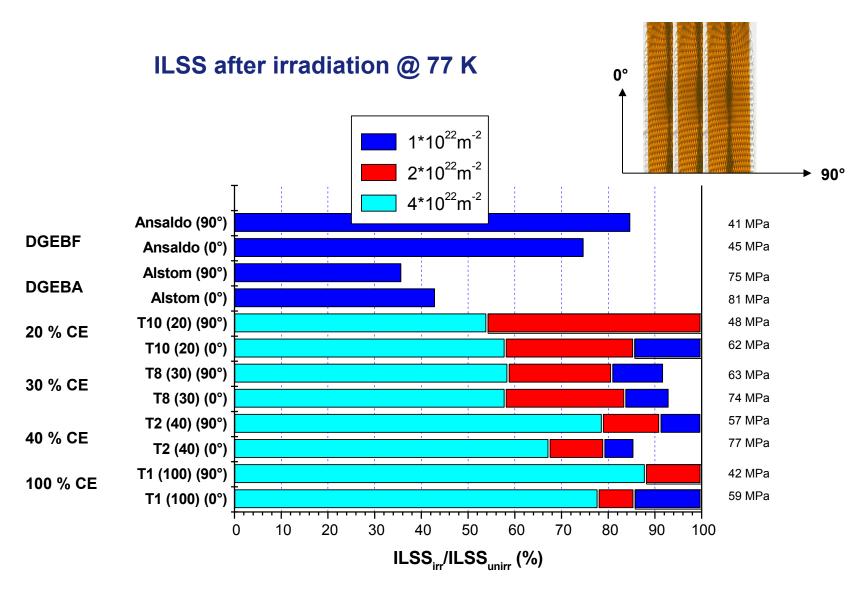








#### Influence of CE content









# **SUMMARY and CONCLUSIONS**

- LT Superconductors: No problems regarding radiation effects expected for ITER
- Stabilizer: Degradation must be kept in mind
- Insulators: Excellent solution found industrial tests completed; qualification of materials from different suppliers under way







# ACKNOWLEDGEMENTS

Work on the superconductors started at ATI in 1977 and was done partly at Argonne, Oak Ridge and Lawrence Livermore National Laboratories as well as at FRM Garching.

Work on the insulators started in 1983 and in systematic form in 1990.

Many graduate students and post-doctoral fellows have been involved.

Substantial support by the European Fusion Programme (EFDA) is acknowledged.

The contributions of the present ATI crew are gratefully acknowledged.

Senior scientists: M. Eisterer, H. Fillunger, K. Humer, R.K. Maix, F.M. Sauerzopf

Post-docs: R. Fuger, F. Hengstberger, R. Prokopec, M. Zehetmayer Graduate students: T. Baumgartner, M. Chudy, J. Emhofer

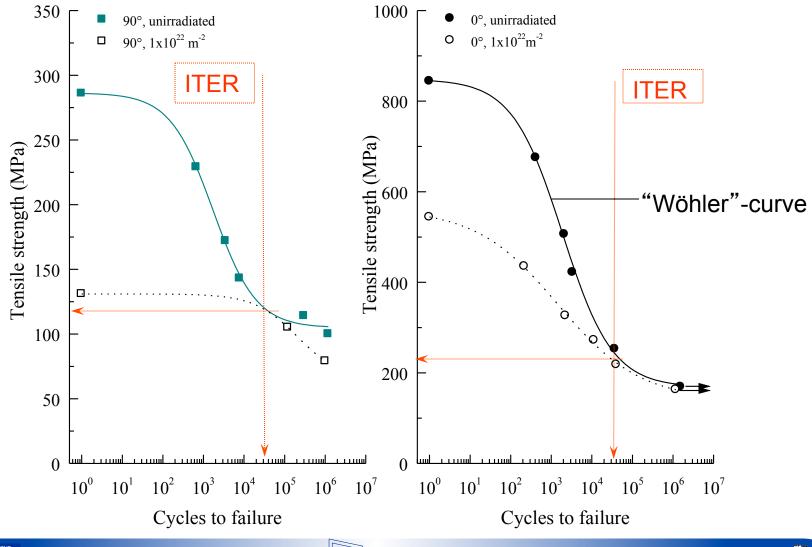






#### EUHT DGEBA Epoxy Ansaldo (similar results for Alstom)

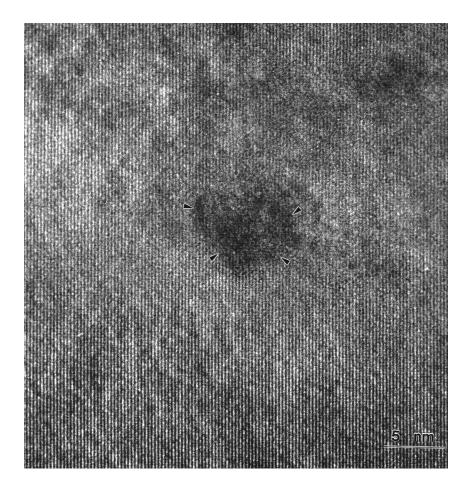
## Tensile fatigue performance @ 77 K











#### STATISTICALLY DISTRIBUTED

~ SPHERICAL, ~ 2.5 nm Ø

SURROUNDED BY A STRAIN FIELD OF THE SAME SIZE

 $5 \times 10^{22}$  defects m<sup>-3</sup> per  $10^{22}$  neutrons m<sup>-2</sup>

#### **FAST NEUTRONS**

COLLISION CASCADES, IF THE ENERGY OF THE PRIMARY KNOCK-ON ATOM EXCEEDS ~ 1 keV







## **Collision Cascade (schematic)**







